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HIGH RESOLUTION HEAT CAPACITY STUDIES NEAR THE NEMATIC-  
SMECTIC A PHASE TRANSITION

*University of Illinois at Urbana-Champaign*

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HIGH RESOLUTION HEAT CAPACITY STUDIES NEAR  
THE NEMATIC-SMECTIC A PHASE TRANSITION

BY

JANE DEBORAH LEGRANGE

B.A., University of Pennsylvania, 1975  
M.S., University of Illinois, 1977

THESIS

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This thesis is dedicated to her father and to the memory of her mother who have always been a source of strength and friendship.

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"Liquid crystals are beautiful and mysterious; I am fond of them  
for both reasons."

P. G. deGennes

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## CHAPTER 1

## INTRODUCTION

Liquid crystals are long, asymmetric organic molecules which do not show a single transition from solid to liquid, but exhibit a variety of phases whose symmetries and mechanical properties are intermediate between those of a liquid and a crystal. These phases are defined by the orientational and spatial ordering of the molecules and can change as a function of temperature (thermotropic liquid crystals) or of concentration of the rodlike molecules in solvent (lyotropic liquid crystals).

Liquid crystal molecules are typically 25 Å long and 5 Å wide. Aromatic rings and adjacent double bond linkages form a rigid core while the alkyl end chains are flexible. The chemical structures of four liquid crystals and the sequence of phases for each are shown in Figure 1.

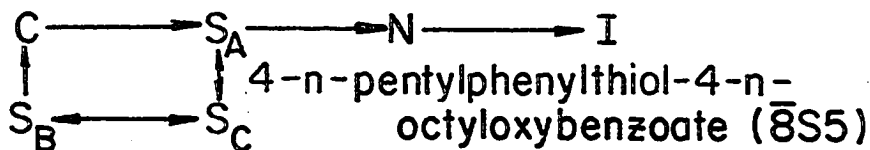
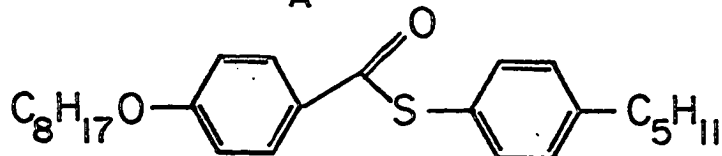
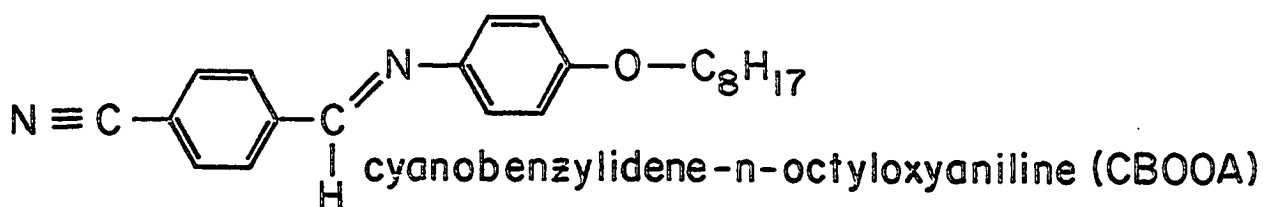
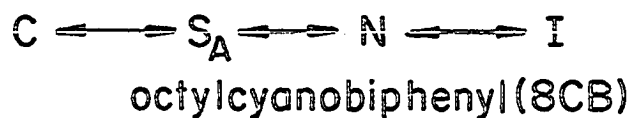
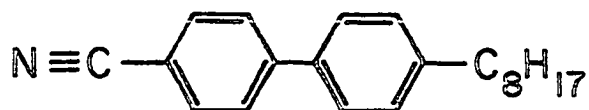
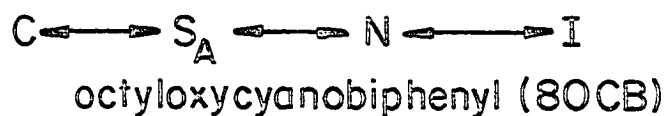
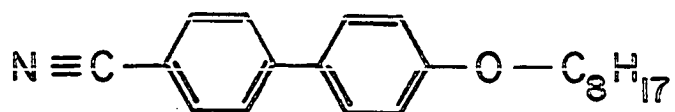
The nematic phase is characterized by long range orientational order where the molecules are aligned along a preferred direction called the director,  $\hat{n}$ , and  $+\hat{n}$  and  $-\hat{n}$  are physically equivalent. There is no order in the positions of the centers of mass. The molecules are free to rotate about their long axes and there is no average dipole moment. The system is optically uniaxial with the optical axis parallel to the director.

In the smectic phases, known as A, B, and C, the molecules are stacked in two dimensional layers. Within each layer there is high orientational order with the molecules aligned along a preferred direction.

Smectics A and C are known as disordered smectics, which can be visualized as stackings of two dimensional liquids. There is no long range



Figure 1. The chemical structure and sequence of phases for the compounds studied.



positional order within the layers in either of these phases. In smectic A the molecules are aligned perpendicular to the layers so that there is one dimensional positional order. The system is optically uniaxial and the molecules are rotationally symmetric about the long axis. The smectic C phase, however, is optically biaxial because of a tilt angle between the normal to the layers and the director.

The smectic B phase is an ordered smectic phase where there is positional ordering and each layer forms a two dimensional solid. Molecules form a lattice within each layer but there is no long range correlation between layers.

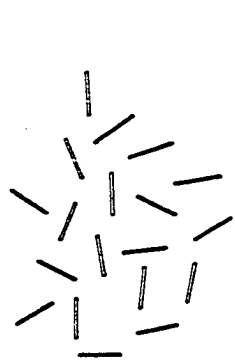
The isotropic liquid phase occurs at a higher temperature than the smectic and nematic phases. As in any liquid there is no long range orientational or positional order. Schematic pictures of these liquid crystalline phases are shown in Figure 2.

Liquid crystals, with their myriad of phase transitions, offer an opportunity to study phase transition phenomena at accessible temperatures. In particular, the nematic-smectic A phase transition may exhibit a tricritical point or crossovers between different types of critical behavior<sup>1/</sup>. Liquid crystals can also be useful for studying the effect of impurities on critical behavior.

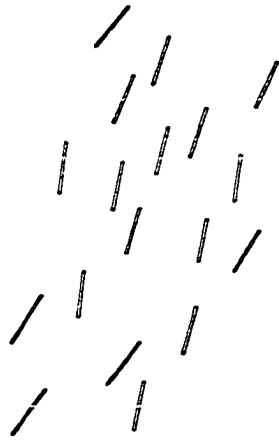
Recently there has been extensive work done on the nematic-smectic A transition in the effort to determine if it can be a continuous (second order) transition and if so which class of critical behavior it exhibits. Several experiments are being done to measure the exponents  $\nu$  and  $\alpha$  whose values provide keys for understanding the microscopic behavior of the liquid crystalline phases, as well as interesting aspects of the phase diagram.

Figure 2. Schematics of several liquid crystalline phases.

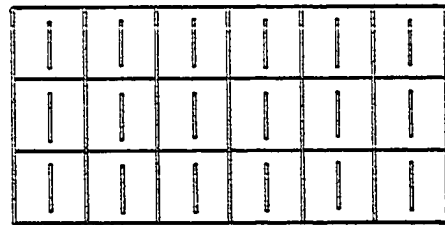
Smectics A and C are liquid-like smectics. B is more ordered with a lattice structure within each layer.



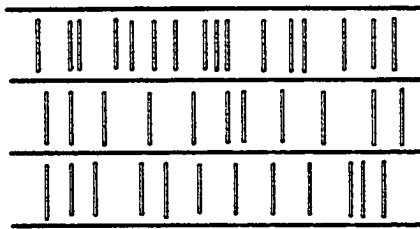
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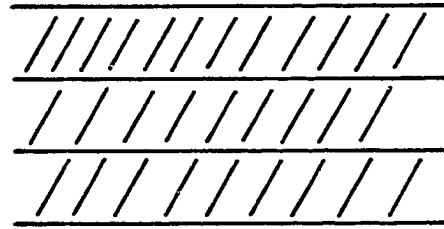
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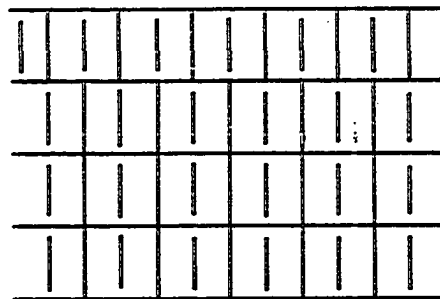
crystal



smectic A



smectic C



smectic B

The a.c. heat capacity method is a technique designed to measure small changes in heat capacity with resolution on the order of a part in  $10^4$  and temperature resolution of 1 mK. The high resolution in the measurement of temperature and heat capacity makes this an ideal method for determining the values of a critical exponent.

This thesis reports careful measurements of heat capacity near the nematic-smectic A transition in several liquid crystal compounds and the results for the exponent and other parameters in the equation for heat capacity at a critical point. Chapter 2 reviews previous theoretical and experimental work done on the nematic-smectic A transition. Chapter 3 describes the experimental techniques and the data fitting procedures. Chapter 4 presents the results for various compounds, including chemical analyses, and concludes the thesis with a discussion of the implications of these results on the understanding of the nematic-smectic A transition.

## CHAPTER 2

THE NEMATIC-SMECTIC A TRANSITION: THEORY  
AND EXPERIMENT2.1 The Order Parameters

The liquid crystalline phases are described by order parameters which can range from 0 to 1 depending on how ordered the system is. In the nematic phase

$$s = \frac{1}{2} \langle 3 \cos^2 \theta - 1 \rangle \quad (2.1)$$

where  $\theta$  is the angle between the long axis of each molecule and the director and  $s$ , the order parameter, is the average of the thermal fluctuations of the molecules around the director.  $S$  varies as a function of temperature and is typically in the range .3 to .8.<sup>2/</sup> In the smectic A phase the order parameter  $\Psi$ , a complex wave function, is a mass density wave

$$\Psi = \Psi_0 \exp(iqz) \quad (2.2)$$

where  $|\Psi|^2$  is the strength of the smectic layering and  $q=2\pi/d$  where  $d$  is the interlayer spacing. Although  $\langle \Psi \rangle = 0$  in the nematic phase, both  $s$  and  $\Psi$  are needed to describe the smectic phase and there is coupling between them. As the smectic layers form it is energetically favorable for the long axes of the molecules to line up perpendicular to the layers.  $S$ , then, is high with a value of .8 or .9.<sup>2/</sup>

## 2.2 Microscopic Mean Field Theory

McMillan<sup>3/</sup> and Kobayashi<sup>4/</sup> have developed molecular mean field theories based on these order parameters and have produced a qualitatively correct phase diagram, shown in Fig. 3, for systems with nematic, smectic A, and isotropic phases. In a given homologous series, where the chemistry of the compounds differs by the length of their alkyl chains, the temperature range of the nematic phase decreases as the alkyl chain increases and at some value of the chain length there is a tricritical point where the transition changes from second to first order. As the molecular length increases beyond the tricritical point the discontinuities in entropy and the order parameters increase. The variable, molecular length, is equivalent to the ratio of the nematic-smectic A transition temperature,  $T_{na}$ , to the nematic-isotropic transition temperature,  $T_{ni}$ . Compounds which exhibit the nematic phase over a wide range in temperature, corresponding to a smaller  $T_{na}/T_{ni}$  ratio, have maximized the orientational order before  $T_{na}$  is reached. There is no discontinuity in either order parameter at  $T_{na}$  and the transition is second order. Compounds with narrow nematic regions, or higher  $T_{na}/T_{ni}$  ratio, do not have their orientational order saturated as they approach  $T_{na}$ . There is a discontinuity in the orientational order parameter  $s$  at  $T_{na}$  as it is energetically favorable for the molecules to line up perpendicular to the smectic layers as they form. The coupling of the order parameters  $s$  and  $\psi$  forces a discontinuity in  $\psi$ . Such discontinuities are characteristic of first order phase transitions, and are observable in compounds whose  $T_{na}/T_{ni}$  ratio is greater than  $T_{na}/T_{ni}$  at the tricritical point. Figure 4 shows plots of the order parameters as functions of temperature for first and second order transitions.



Figure 3. The phase diagram from the mean field theories of McMillan<sup>3/</sup> and Kobayashi.<sup>4/</sup> The nematic-smectic A transition temperature,  $T_{na}$ , increases with the length of the alkyl tail chain. At a critical value of chain length there is a non-zero entropy change at  $T_{na}$  and a first order phase transition.

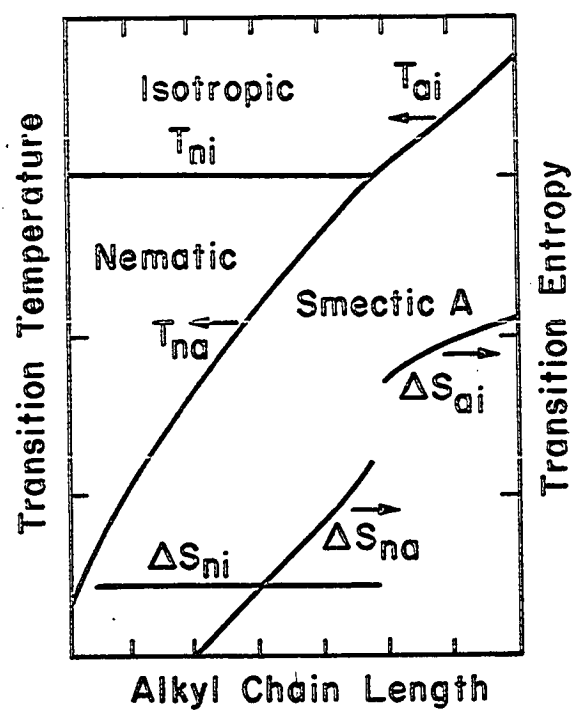
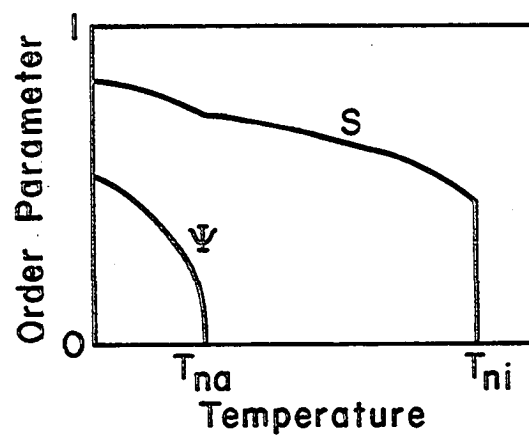
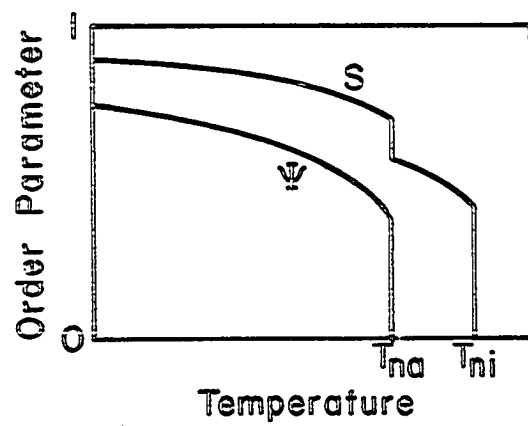
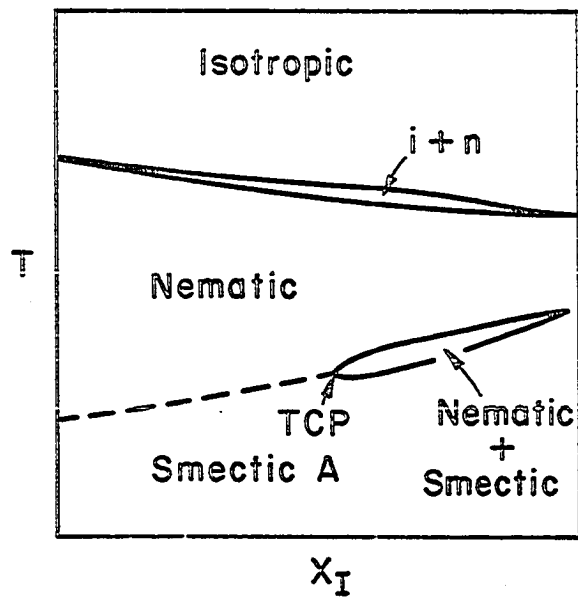


Figure 4. Discontinuities in the coupled order parameters  $s$  and  $\Psi$  as functions of temperature indicate a first order phase transition. A continuous variation in  $s$  and  $\Psi$  at  $T_{na}$  indicates a second order phase transition.



The mean field theories of McMillan<sup>3/</sup> and Kobayashi<sup>4/</sup> are the first to predict the existence of a tricritical point on the phase diagram and the possibility of a second order phase transition. There has been much experimental work aimed at locating the value of  $T_{na}/T_{ni}$  for the tricritical point and at determining if there are second order nematic to smectic A phase transitions. Alben<sup>5/</sup> has suggested that the search for the tricritical point could be carried out by doping a liquid crystal which has a second order nematic-smectic A transition with different amounts of a higher homolog which undergoes a first order transition, leading to a continuous variation of effective  $T_{na}/T_{ni}$  with concentration. The proposed phase diagram for such a series of mixtures is illustrated in Figure 5 and resembles that of  $\text{He}^3\text{-He}^4$  mixtures. That portion of the phase diagram where the concentration of the longer homolog is greater than the concentration at the tricritical point includes a coexistence region between the nematic and smectic A phases. Various groups doing measurements of latent heats on such mixtures have found the tricritical point in the range  $T_{na}/T_{ni} = .85$  to  $.96$ .<sup>6/</sup> Many experiments have also been done to determine the order of the NA transition in pure CBOOA, which is believed to be near the tricritical point with a value of  $.94$  for  $T_{na}/T_{ni}$ . McMillan<sup>7/</sup> does not observe a latent heat in the NA transition of CBOOA and Cabane and Clark,<sup>8/</sup> in measuring  $s$  as a function of temperature using nuclear quadrupole resonance, do not observe a discontinuity in the order parameter. Both of these measurements indicate a second order NA transition. Contradicting these results, Torza and Cladis<sup>9/</sup> measure a slight volume discontinuity and Djurek et al.<sup>10/</sup> measure small latent heats at  $T_c$  indicating a first order transition. These varying results for CBOOA

Figure 5. A proposed phase diagram for mixtures of liquid crystals resembling the phase diagram for  $\text{He}^3\text{-He}^4$  mixtures.  $x_I$  is the concentration of the liquid crystal with a longer end chain. The average length of the end chain increases with greater concentrations of the longer homolog. At a certain average length the tricritical point is reached and there is a coexistence region between the nematic and smectic A phases, indicating a first order transition.



may be explained by the nearby presence of the tricritical point where chemical differences such as small concentrations of shorter molecules, either homologs or broken segments of CBOOA, can determine whether the sample exhibits first or second order behavior. The effects of chemical impurities on liquid crystal phase transition physics are discussed in detail in Chapter 4.

### 2.3 The Landau Free Energy and Analogy with a Superconductor

DeGennes<sup>11/</sup> and McMillan<sup>12/</sup> have proposed a Landau theory where the free energy incorporates the coupling of the smectic order parameter to the nematic director.

$$F_1 = \int d\vec{r} \left[ A(T) |\Psi|^2 + \frac{B}{2} |\Psi|^4 + C_{||} |(\vec{n} \cdot \vec{\nabla} - iq)\Psi|^2 + C_{\perp} |\vec{n} \times \vec{\nabla} \Psi|^2 \right]. \quad (2.3)$$

The  $C_{||}$  term expresses the cost in energy of compressing the smectic layers and the  $C_{\perp}$  term represents the cost of rotating the director within the layers. Only even powers appear in the free energy as it must be invariant under changes in the phase of  $\Psi$ . The coefficient  $A$  is positive in the nematic phase and goes to zero at  $T_{na}$ . The sign of the coefficient  $B$  determines the order of the phase transition.

$$B = B_0 - 1/2C^2\chi \quad (2.4)$$

where  $C$  is a constant combining  $C_{||}$  and  $C_{\perp}$  from the coupling terms,  $B_0$  is positive, and  $\chi$  is a response function which is large and positive when there are large fluctuations in the order parameter  $s$ . If the order parameter  $s$  is saturated near  $T_{na}$ , as in the case of a wide nematic phase,  $\chi$  is small so  $B$  is positive, and the transition is second order. If  $\chi$  is large enough to make  $B$  negative, as in compounds with a narrow nematic



region, then a  $\Psi^6$  term must be added to ensure stability and the resulting plots of  $F(\Psi)$  show that at  $T_{na}$   $F$  is minimized at a non-zero  $\Psi$ , and the transition actually occurs at a higher temperature,  $T_c > T_{na}$ , and is first order.

DeGennes<sup>13/</sup> has noted that this expression resembles the Ginsberg-Landau free energy of a superconductor where the coupling between the director and smectic order parameter is analogous to the coupling of the magnetic vector potential with the order parameter. The order parameter in both the superconductor and liquid crystal has two components, the amplitude and phase. In the superconductor the zero temperature coherence length is hundreds to thousands of angstroms long causing mean field like behavior. In a liquid crystal, where the interactions are short ranged and the coherence length is on the order of a molecular length, helium-like behavior is expected. Based on this analogy DeGennes<sup>11/</sup> predicts the helium values,  $\nu=.66$  and  $\alpha=0$ , for the critical exponents near the nematic-smectic A phase transition. DeGennes has also predicted that the smectic A phase excludes the bend and twist distortions described below, as a superconductor excludes magnetic flux.

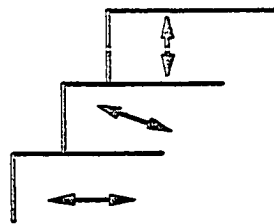
In Figure 6 the twist, bend, and splay distortions are illustrated. The bend and twist modes are excluded in the smectic phase as such distortions break up the interlayer spacing and cost energy. A splay distortion maintains the interlayer spacing and does not cost as much energy. The following terms which describe these fluctuations of the director are included in the free energy,

$$F_2 = \int d\vec{r} \left[ \frac{1}{2} K_{11} (\vec{\nabla} \cdot \vec{n})^2 + \frac{1}{2} K_{22} \left( \vec{n} \cdot (\vec{\nabla} \times \vec{n}) \right)^2 + \frac{1}{2} K_{33} \left( \vec{n} \times (\vec{\nabla} \times \vec{n}) \right)^2 - \chi_a (\vec{n} \cdot \vec{H})^2 \right], \quad (2.5)$$

Figure 6. A schematic illustration of the splay, bend, and twist fluctuations.



Splay ( $\text{div } \vec{n} \neq 0$ )



Bend ( $\text{curl } \vec{n} \perp \vec{n}$ )



Twist ( $\text{curl } \vec{n} \parallel \vec{n}$ )

where  $F_{\text{total}} = F_1 + F_2$ . The  $K_{ii}$  are the elastic constants (splay, twist, and bend) and the last term lowers the free energy when the molecules are aligned in a magnetic field.

## 2.4 Critical Exponents

The thermal average of  $\langle \Psi \rangle$  is positive in the smectic phase, decreases as temperature approaches  $T_c$ , ( $T_c = T_{na}$  for a second order phase transition), and drops to zero in the nematic phase. Although  $\langle \Psi \rangle$  vanishes in the nematic phase there are fluctuation effects whereby short range order persists over a distance  $\xi(T)$  larger than the molecular length. Droplets with smectic A ordering whose dimensions are the components of the correlation length,  $\xi(T)$ , parallel and perpendicular to the director, form in the nematic phase and increase in size as  $T_c$  is approached from above.

$$\xi_{\perp} \propto \left[ (T - T_c) / T_c \right]^{-\nu_{\perp}} \quad (2.6)$$

$$\xi_{\parallel} \propto \left[ (T - T_c) / T_c \right]^{-\nu_{\parallel}} \quad (2.7)$$

where  $\nu_{\perp}$  and  $\nu_{\parallel}$  are non-integral exponents. As the smectic fluctuations are enhanced it costs more energy to bend or twist the director. This is expressed in the renormalization of the elastic constants which grow as functions of  $\xi_{\parallel}$  and  $\xi_{\perp}$ .

The growth of the smectic fluctuations results in the increase of other physical quantities near  $T_c$  as functions of reduced temperature,  $t$ , raised to a non-integral exponent, where  $t = (T - T_c) / T_c$ . The heat capacity near a critical point is proportional to  $t^{-\alpha}$  where  $\alpha$  is a critical exponent and is the same on both sides of the transition. There exist several

classes of phase transitions characterized by the values of the critical exponents. A correct determination of critical exponents near the nematic-smectic A transition could place this transition in a universality class and provide better understanding of the physics near  $T_c$  using our knowledge of other systems in that class.

There are only a few groups measuring critical exponents on the nematic-smectic A transition and the results of their experiments vary considerably. Johnson et al., in an a.c. heat capacity experiment, have measured  $\alpha$  in pentylphenylthiol-octyloxybenzoate (8S5)<sup>14/</sup> and in octyloxy-cyanobiphenyl (8OCB)<sup>15/</sup> and find  $\alpha$  near zero indicating the logarithmic behavior of a superfluid-like transition. In their 8OCB measurements they observe a 40 mK flattened region at the transition. This could be a two phase region which is characteristic of a first order phase transition. In another a.c. heat capacity experiment Garland<sup>16/</sup> has measured  $\alpha$  between .2 and .3 on 8OCB and also on octylcyanobiphenyl (8CB). Birgeneau<sup>17,18/</sup> at MIT has used light scattering and high resolution x-ray scattering to obtain values for  $\nu_{||}$  and  $\nu_{\perp}$  for 8OCB, 8CB, and CBOOA (cyanobenzoyloxyoctylaniline). Using the anisotropic version of the Josephson scaling law,<sup>19/</sup>

$$2\nu_{\perp} + \nu_{||} = 2 - \alpha, \quad (2.8)$$

the exponents  $\nu_{\perp}$ ,  $\nu_{||}$ , and  $\alpha$  are related to one another. In Table 1, Birgeneau's results on the various compounds are shown as well as the values for  $\alpha$  obtained from the Josephson scaling law by inserting his measured values of  $\nu_{||}$  and  $\nu_{\perp}$ . 8OCB is the one compound that has been measured by all three of these groups resulting in values for  $\alpha$  of 0 (Johnson),<sup>15/</sup> .25 (Garland),<sup>16/</sup> and .13 (Birgeneau).<sup>17,18/</sup> This thesis reports the results

of our high resolution a.c. heat capacity measurements on 80CB, 8CB, CB00A,  $\overline{8S5}$ , and CBNOA and attempts to clarify the confusion surrounding the nematic-smectic A transition.

TABLE 1  
X-RAY RESULTS FOR  $v_{||}$  AND  $v_{\perp}$  AND CALCULATED  $\alpha$

Compound	$v_{  }$	$v_{\perp}$	$\alpha$
CB00A	$.70 \pm .04$	$.62 \pm .05$	$.06 \pm .14$
80CB	$.71 \pm .04$	$.58 \pm .04$	$.13 \pm .12$
8CB	$.67 \pm .04$	$.51 \pm .04$	$.31 \pm .12$

## CHAPTER 3

## HEAT CAPACITY EXPERIMENTS

3.1 The A.C. Method

The a.c. calorimeter used for the measurements is a modification of the one originally designed by J. E. Smaardyk,<sup>20,21/</sup> and is based on the a.c. heat capacity technique of Sullivan and Seidel.<sup>22/</sup> An oscillating voltage is applied across a thin film heater in contact with the sample. The resulting temperature oscillations, which are inversely proportional to heat capacity, are measured as a function of temperature by a thin film thermocouple underneath the sample.

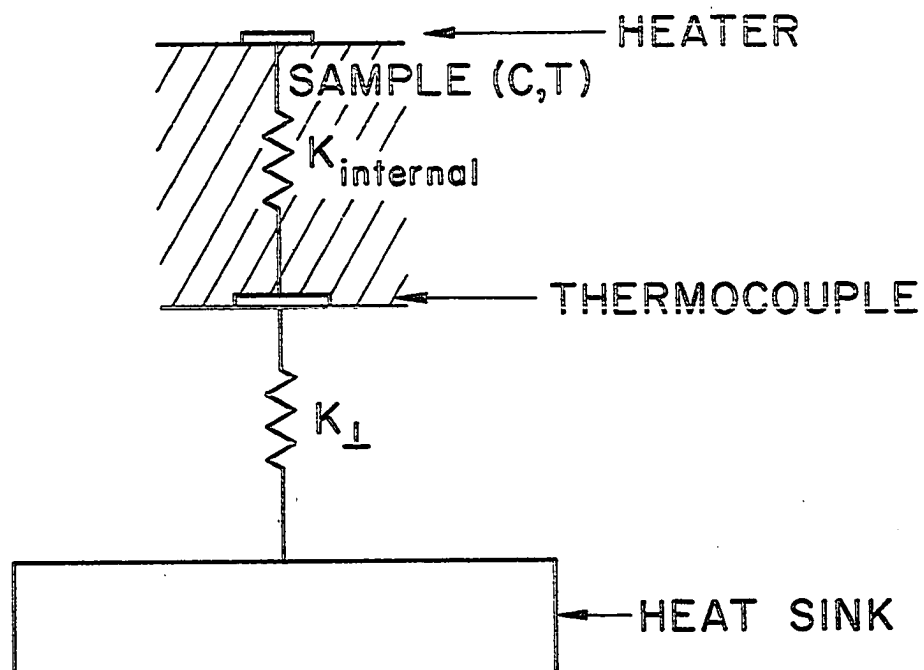
A schematic model of an a.c. calorimeter is shown in Figure 7. The heat flow from the thin film heater on the sample, meaning the heater-liquid crystal assembly, to the heat sink has radial and perpendicular components. By adding exchange gas behind the window as a thermal link between the sample cell and heat sink, the perpendicular thermal conductivity becomes much greater than the radial thermal conductivity. The change in the energy of the sample complex can be written as follows:

$$C \frac{dT}{dt} = \frac{V^2(t)}{R} - K_{\perp}(T - T_0) \quad , \quad (3.1)$$

where C is the heat capacity of the sample,  $K_{\perp}$  is the thermal conductance from the sample cell through the exchange gas underneath, R is heater resistance, and  $T_0$  is the temperature of the heat sink or thermal bath. If

Figure 7. Schematic model of a.c. calorimetric technique.  $K_{\text{internal}}$  is the thermal conductivity through the sample complex from the thin film heater to the thermocouple.  $K_{\perp}$  is the perpendicular component of thermal conductivity from the thermocouple through the gas to the bath. The exchange gas ensures that the perpendicular components of thermal conductivity are much greater than the parallel components. The heat capacity measured,  $C$ , is the heat capacity of the entire sample assembly.





$V(t)$  is a sinusoidal voltage of amplitude  $V_0$  and frequency  $\omega/2$ , the equation can be solved for  $T$  where

$$T = T_0 + V_0^2/2R \left[ \frac{1/K_{\perp} + 1/\omega C}{1 + 1/\omega^2 \tau_{\text{ext}}^2} \sin(\omega t + \phi) \right]$$

$$= T_0 + T_{\text{dc}} + T_{\text{ac}} \sin(\omega t + \phi) \quad (3.2)$$

$\tau_{\text{ext}} = C/K_{\perp}$  is the thermal relaxation time of the sample to the bath and  $\phi = \arctan 1/\omega \tau_{\text{ext}}$ . The dc temperature rise of the sample,

$$T_{\text{dc}} = V_0^2/2R K_{\perp} \quad (3.3)$$

is the averaged difference between the sample temperature and the heat sink temperature where  $V_0^2/2R$  is the average power. If  $\omega \tau_{\text{ext}} \gg 1$  then  $T_{\text{ac}}$ , the amplitude of the a.c. component of temperature, simplifies to:

$$T_{\text{ac}} = V_0^2/2R \omega C \quad (3.4)$$

and the heat capacity  $C$  can be calculated from the amplitude of the temperature oscillation. This condition defines a lower limit on the frequency range for the a.c. technique.

The upper limit of this range is defined by the internal relaxation time:

$$\tau_{\text{in}} \ll 1/\omega \quad (3.5)$$

in other words, the sample must not have internal temperature gradients.

$\tau_{\text{in}} = C/K_{\text{in}}$  where  $K_{\text{in}}$  is the thermal conductance from the thin film heater on top of the sample cell to the thin film thermocouple on the glass. The

thermal wavelength associated with the internal relaxation time,

$$\ell = \sqrt{k_{in}/\omega c} \quad , \quad (3.6)$$

where  $k_{in}$  is thermal conductivity and  $c$  is heat capacity per unit volume of the sample complex, is the length which determines how thick the sample cell can be so that all parts of it oscillate in phase.

Before each measurement a frequency response curve is plotted to find the range of frequency where heat capacity is not a function of frequency.  $C$  is inversely proportional to  $\omega T_{ac}$  so by plotting  $\omega T_{ac}$  against  $\omega$ , a flat frequency response is obtained whose limits are determined by the reciprocals of the internal and external relaxation times described above. The a.c. calorimeter must be operated in this frequency range. An example of such a frequency response curve is shown in Figure 8.

### 3.2 The Liquid Crystal Sample Cell

A schematic of the liquid crystal sample cell is shown in Figure 9. A pyrex window, several microns thick, is epoxied onto a copper mount with Stycast 1266.<sup>23/</sup> The copper mount is 3 cm. in diameter and in its center is a hole with a diameter of 0.5 cm. It is important for the support of the glass that the epoxy reach the edge of the hole. The amplitude of the temperature oscillation is measured with a thin film copper-bismuth thermocouple. A 7K Å thick film of bismuth is evaporated from a tungsten strip onto the glass and then, while still under vacuum, the copper mount covered by the pyrex window is rotated 90° over the mask and a 2.5K Å film of copper is deposited across the bismuth from a tungsten strip. This method ensures that no oxides form between the metal layers. The configuration of the thermocouple junctions on the glass is depicted in Figure 10.

Figure 8. The flat region of the frequency response curve determines the operating range for the a.c. technique. Heat capacity is inversely proportional to  $f\chi T_{ac}$ . The operating range is determined by the reciprocal of the internal relaxation time  $1/\tau_{int}$  and the reciprocal of the external relaxation time  $1/\tau_{ext}$ .

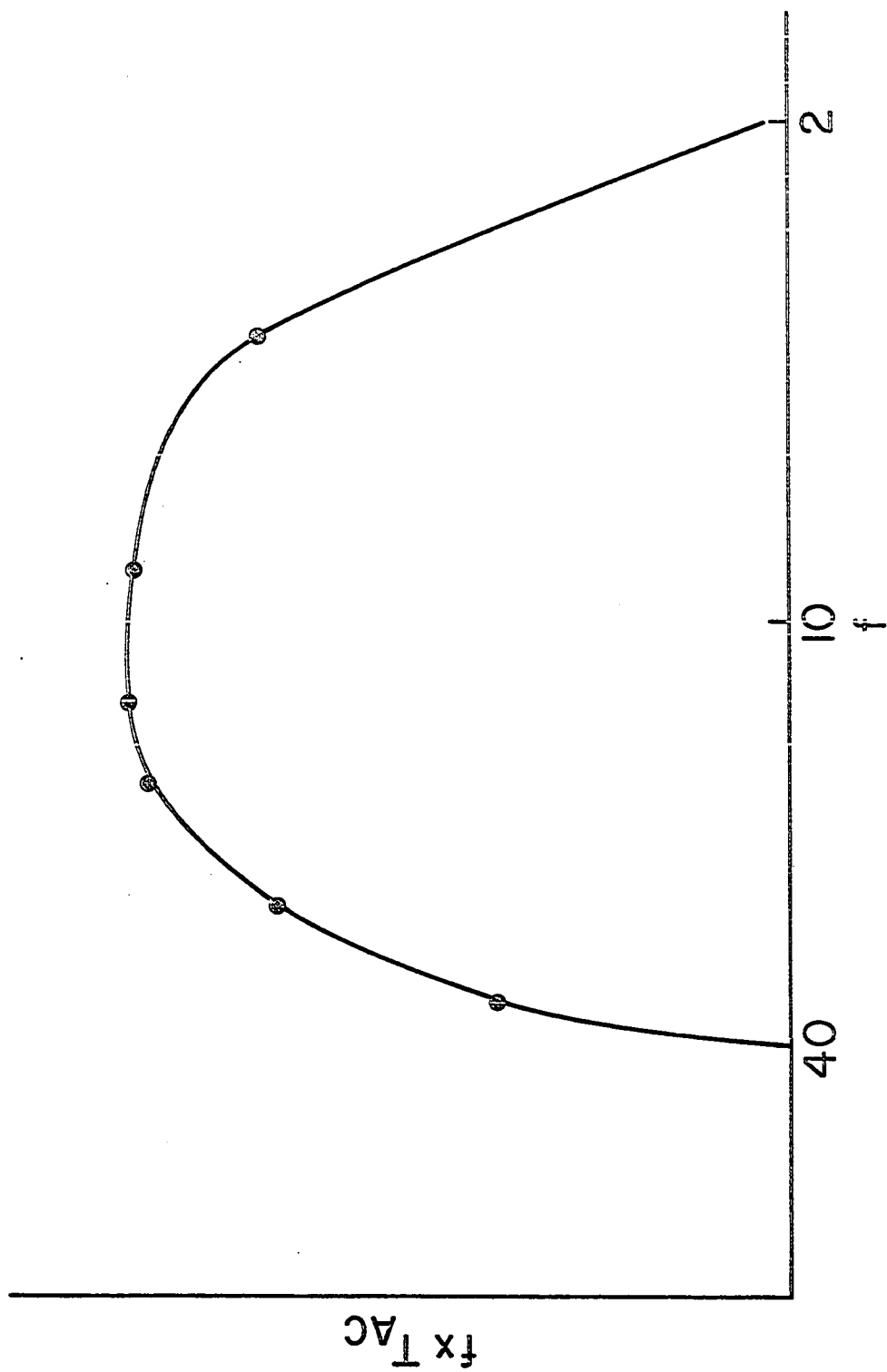


Figure 9. The experimental cell for the a.c. heat capacity measurements of liquid crystal. A pyrex window with a thin film thermocouple junction in its center is epoxied over a hole in a copper disk. The liquid crystal lies between the pyrex window and a sheet of mylar with a thin film heater on its top. Several torr of exchange gas behind the window ensures perpendicular heat flow from the thin film heater.

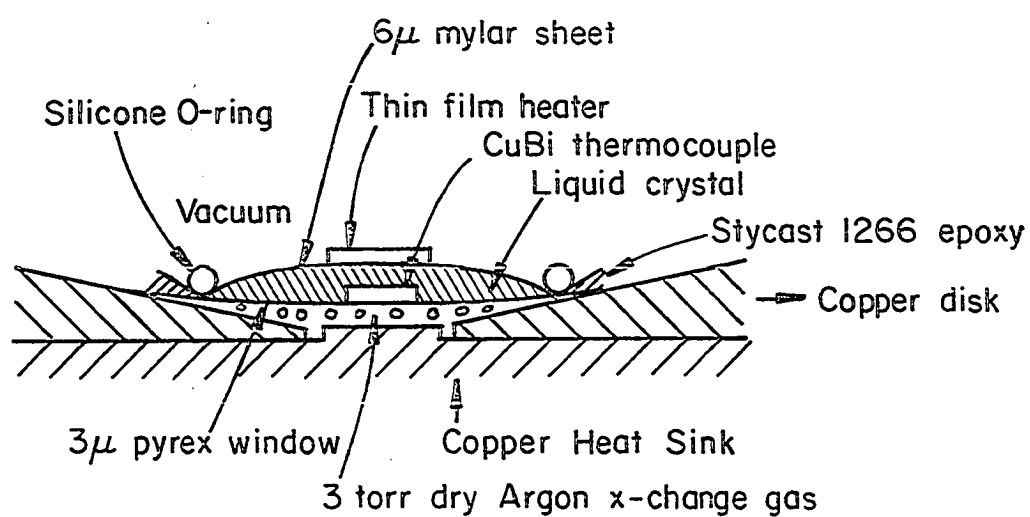
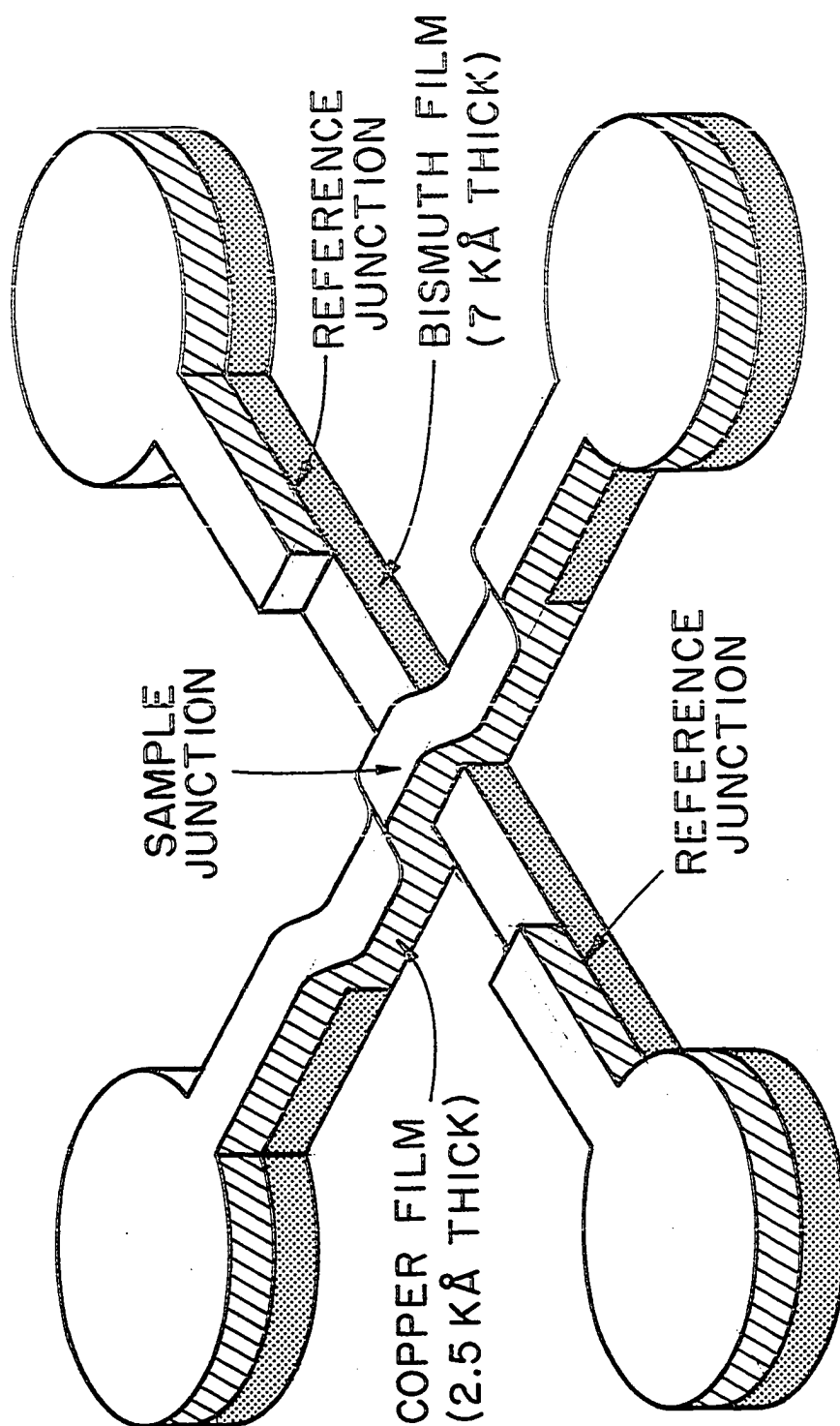


Figure 10. Configuration of bismuth-copper thermocouple used to measure temperature oscillations of liquid crystal.

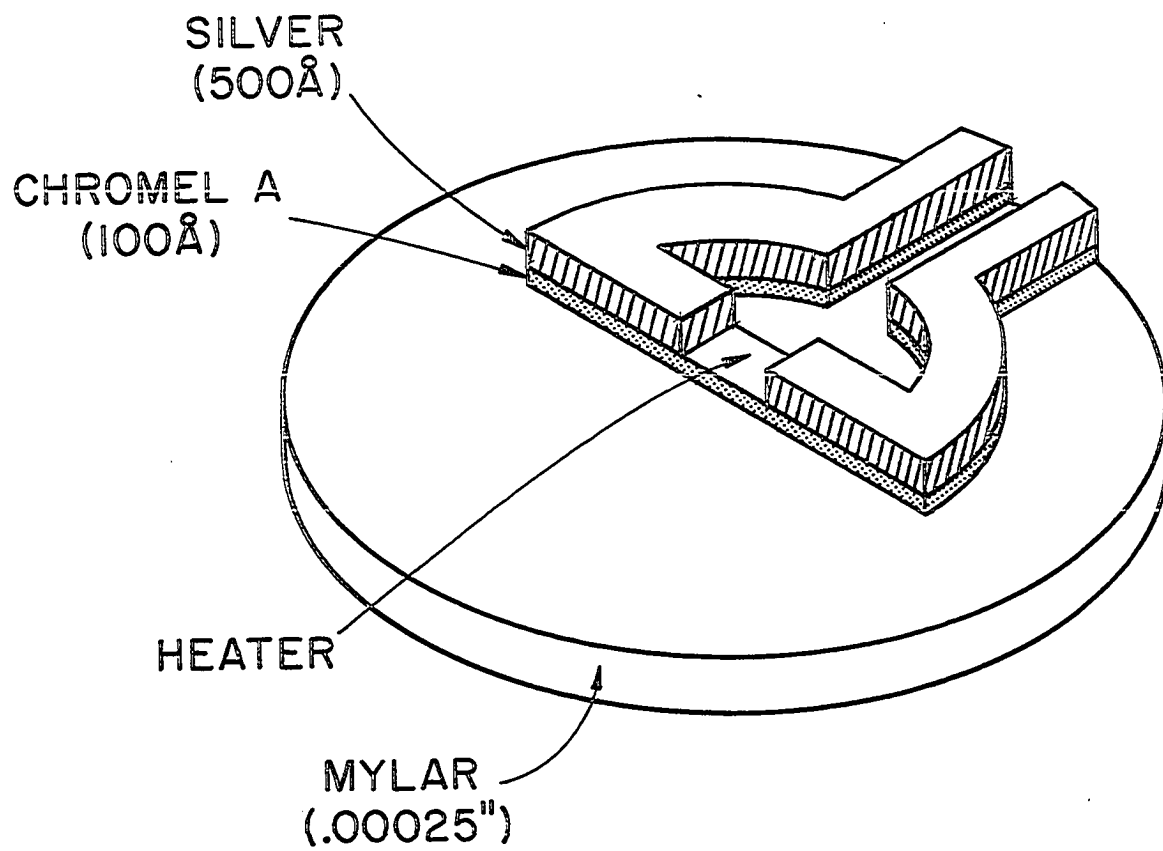




The resistance between the junctions is about  $10\ \Omega$  and the thermoelectric power is on the order of  $50\ \mu\text{V}/^\circ\text{K}$ . Attached to the glass is a  $6\ \mu$  thick, 1.27 cm diameter mylar<sup>24/</sup> disk on which are evaporated the heater and its leads. The heater is a  $2\ \text{mm} \times 2\ \text{mm}$  square of Chromel A,<sup>25/</sup>  $100\ \text{\AA}$  thick, and the leads are thin silver films on the order of  $500\ \text{\AA}$  thick. These films are evaporated from tungsten strips in the configuration shown in Figure 11 during the same vacuum cycle to prevent oxidation between the metals. This is done by supporting the mylar disk over a wheel containing the masks, which is rotated between evaporations to bring the proper mask into position. The Chromel A film extends under the silver leads for better adhesion between the films. The heater resistance ranges from 100-500  $\Omega$  depending on the actual thickness of the Chromel A film. The thicknesses of all the metal films are monitored by a Kronos film thickness monitor attached to a quartz crystal oscillator inside the evaporator.

The mylar is attached to the glass with the heater film on top using a drop of Torr Seal<sup>26/</sup> between the heater leads. The copper mount with the mylar on it is placed underneath a weight while the Torr Seal is applied. It is important that the thermocouple junction be directly underneath the heater so that there will be no temperature gradient across it. Silver wires, .002" in diameter and 99.99% pure, are attached to the edges of the silver films under microscope with small dabs of silver epoxy<sup>27/</sup> on the silver films and on the glass. 10 mil silver wires (99.95% pure) are then attached to the tabs of the thermocouple junctions and to the end of the 2 mil silver wires from the heater films using more silver epoxy. The silver epoxy provides electrical contact with lower contact resistance when used with silver wire. These silver wires are further supported by

Figure 11. Configuration of thin film heater. Heater is a  $2\text{ mm} \times 2\text{ mm}$  square of Chromel A,  $100\text{ \AA}$  thick, and the leads are  $500\text{ \AA}$  thin silver films.



dabs of Torr Seal holding them on the outer edge of the pyrex window. A top view of the completed sample mount is shown in Figure 12.

The sample mount is bolted onto a heat sink (the thermal bath), which sits on three stainless steel pins in a vacuum can. There are two leads from the reference and center junctions and each pair is twisted in parallel to minimize the noise. The reference junction is grounded using silver epoxy. The silver leads are soldered to #32 copper wires going to the outside of the vacuum can. A copper button, which is slightly smaller than the hole in the copper mount protrudes from the heat sink, and is lined up directly underneath the thermocouple junction for better perpendicular thermal conductivity. After the vacuum can is pumped out several torr of dry argon gas are let in behind the window. A cross-sectional view of the assembled calorimeter is shown in Figure 13.

Sealing the liquid crystal between the mylar and glass is the most difficult aspect of the experiment. The measurements are of small changes in heat capacity, which are meaningful only if there are no small drifts in the amount of sample sealed. The clamping device, which is used to seal the sample cell, is illustrated in Figure 14. A silicone O-ring,<sup>28/</sup> .01" in diameter, is dropped onto the mylar under vacuum and held in place by a screw pushing against the brass button holding the O-ring. In addition, several drops of Stycast 1266 are added around the edge of the mylar to ensure that no liquid crystal will seep out from underneath the O-ring.

### 3.3 The Liquid Crystal Samples

The samples measured were all shown in Figure 1. The first experiments were done on octyloxycyanobiphenyl (8OCB),<sup>29/</sup> which was chosen for

Figure 12. Top view of sample mount. Silver epoxy is used for mechanical and electrical contact between heater and thermocouple films and silver wires. Torr Seal is used to adhere the mylar disk to pyrex.

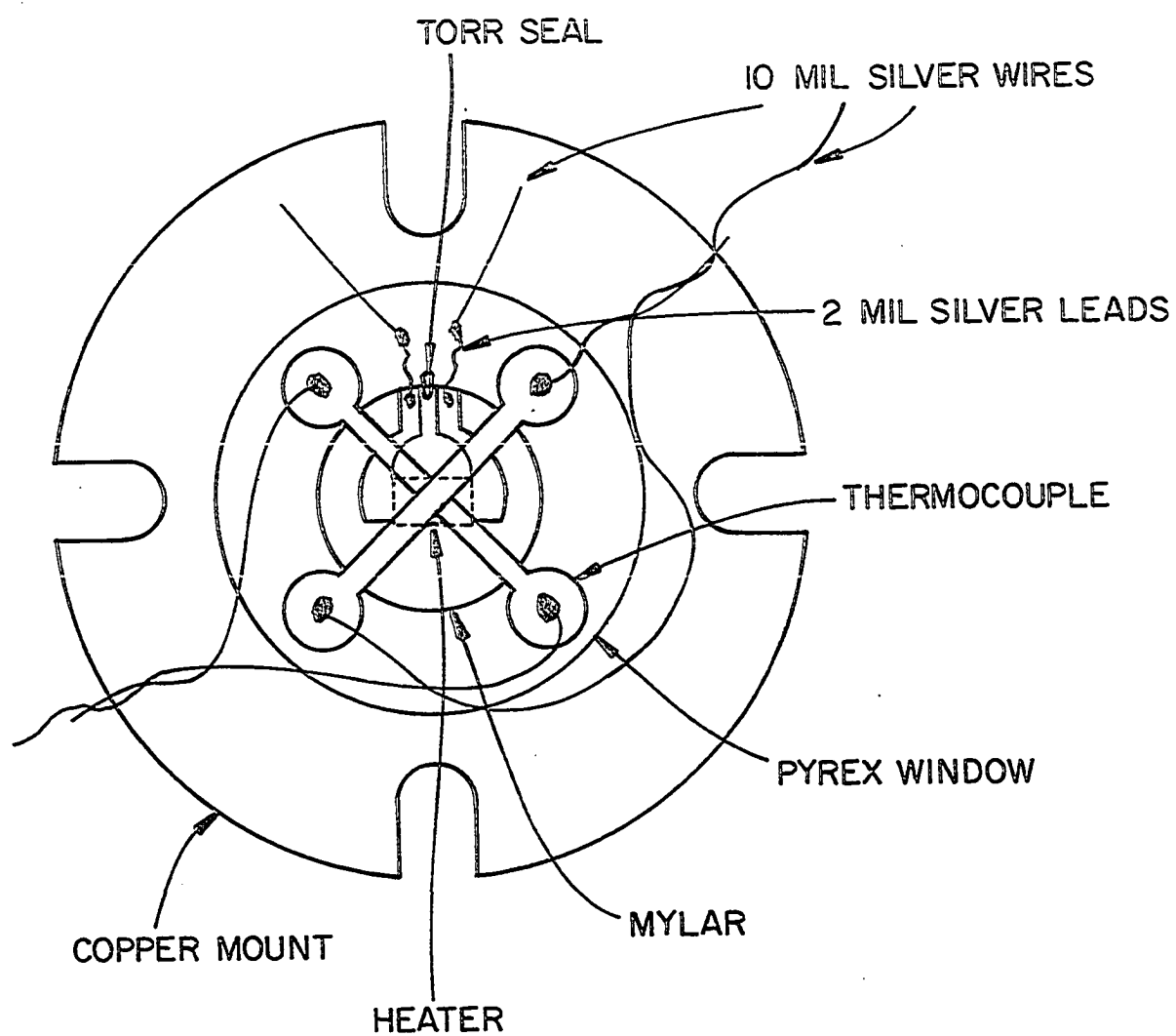


Figure 13. Cross-sectional view of assembled calorimeter. Copper sample mount is bolted onto copper heat sink. Protruding copper button and exchange gas behind window ensure better perpendicular thermal conductivity.



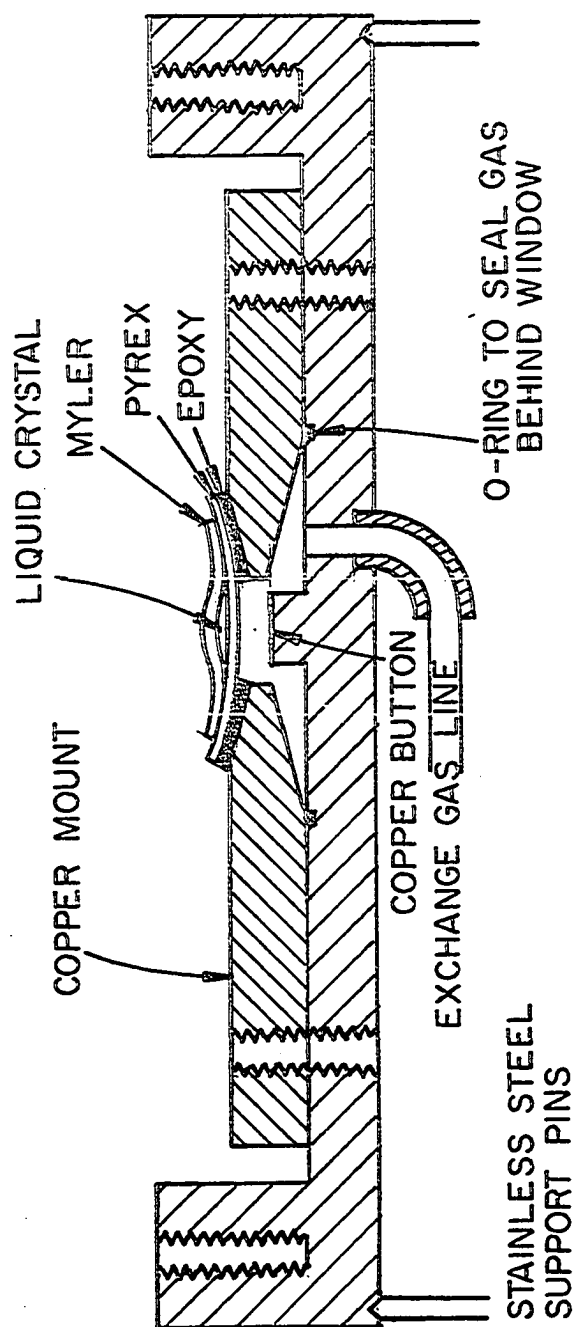
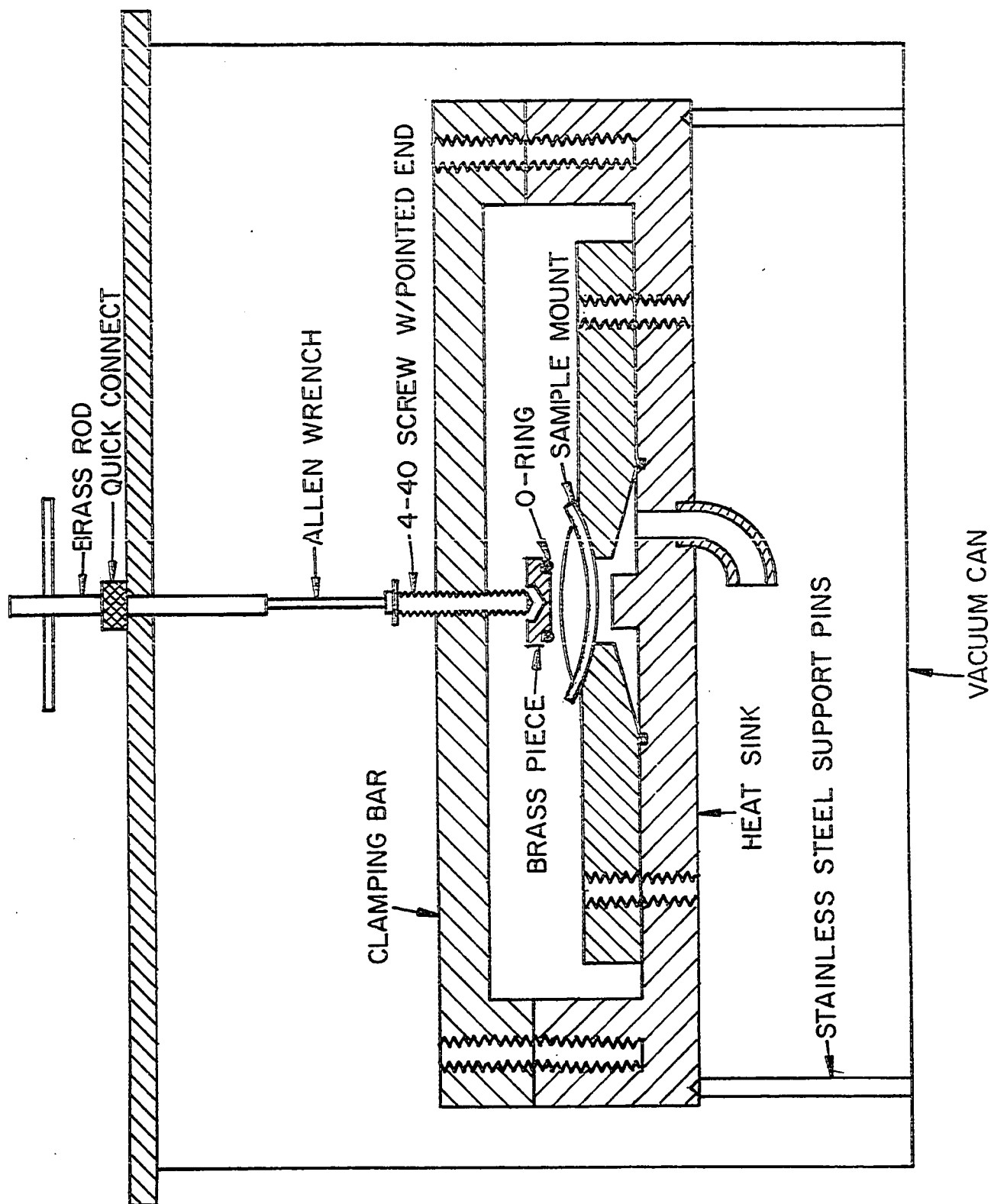


Figure 14. Illustration of clamping procedure to seal liquid crystal. O-ring on the brass piece is held onto the clamping bar with a brass clamp. The pointed screw is engaged from the outside and turned until the O-ring drops into place on the mylar.



its chemical stability. The measurements were done on two different samples from different batches of 80CB made by British Drug Houses.<sup>30/</sup> The first sample came from the earliest batch made by BDH. It was sent to us by P. E. Cladis and had been further purified by vacuum sublimation. The second sample, given to us by D. L. Johnson, was from a later BDH batch and had not been further purified.

Octylcyanobiphenyl (8CB), also made by BDH, was sent to us by D. L. Johnson without further purification.

4-n-pentylbenzenethio-4'-n-octyloxybenzoate ( $\overline{8S5}$ ) was sent to us by D. L. Johnson and had been recrystallized five times from ethanol by Mary Neubert and T. A. Santors.

Cyanobenzylidene-octyloxyaniline (CBOOA) was obtained from Eastman<sup>31/</sup> (compound #11963). 500 mg of CBOOA were recrystallized from 1 ml benzene and 7 ml of hexane. The solution was refluxed, then cooled to recrystallize the CBOOA. The material was filtered and the above process repeated several times. The sample was melted and recrystallized to separate off the solvent.

Cyanobenzylidene-nonyloxyaniline (CBNOA), a higher homolog of CBOOA with a nine carbon tail, was synthesized by Smaardyk. Approximately 500 mg of this batch were recrystallized three times in ~20 ml of ethanol. The recrystallization procedure was the same as described for CBOOA.

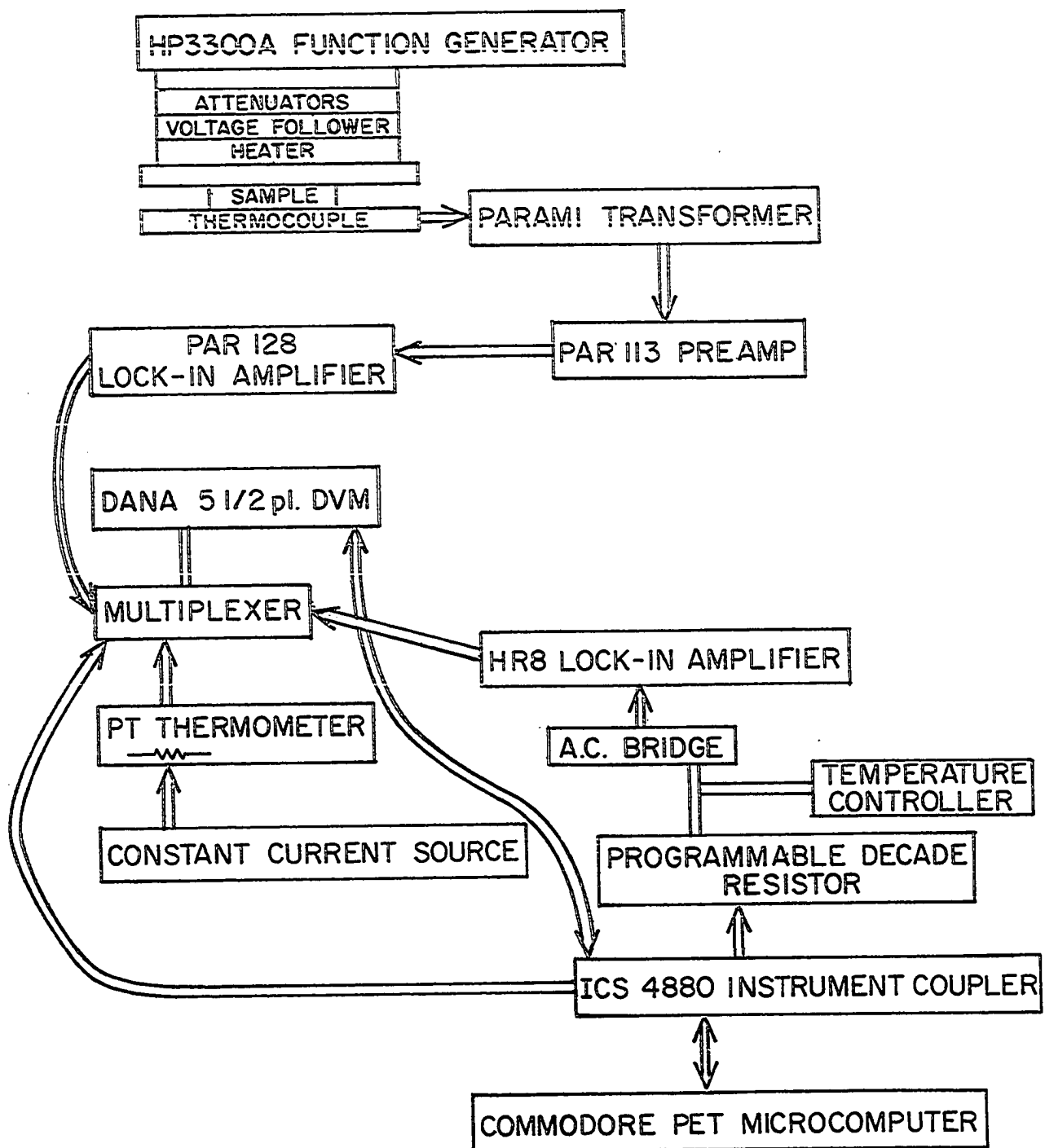
All samples were stored in a dessicator and handled with clean tweezers. The calorimeter was baked out at 90° C for several hours before any sample was added. After being placed in the calorimeter each sample was heated under vacuum to a temperature above the melting point and was pumped on for about thirty minutes before it was sealed. This process should have driven off any volatile impurities. A cold trap filled

with liquid nitrogen protected the unsealed sample from oil vapors from the diffusion pump.

### 3.4 The Electronics and Measuring Techniques

A PET microcomputer<sup>32/</sup> is used to control and measure the amplitude of the temperature oscillations. The root-mean-square amplitude of these oscillations is varied from 1 to 10 mK depending on the resolution desired. A block diagram of the electronics used to measure the temperature oscillations of the sample is shown in Figure 15. A Hewlett Packard Model 3300A function generator supplies the sinusoidal voltage for the thin film heater and the reference input for the lock-in measuring the temperature oscillations. This voltage is attenuated and passes through a voltage follower and a resistor in series with the heater, whose value is equal to the heater resistance. The rms average of the power across the heater ranges from 20 to 400  $\mu$  watts. The d.c. temperature rise of the sample above the heat sink is calculated by plotting the apparent transition temperature for the nematic-smectic A transition as a function of heater power and extrapolating to zero heater power. The a.c. temperature difference between the heat sink and sample corresponds to the a.c. voltage between the reference and center junctions of the thermocouple. For a rms temperature oscillation of 5 mK the voltage between the thermocouple junctions would be several tenths of a microvolt. This voltage is amplified through a P.A.R. AM1 transformer and P.A.R. 113 preamplifier and then measured with a P.A.R. 128 lock-in amplifier in the 2f mode. The total gain of the electronics is roughly a million. The output of the 128 is monitored by a DANA Model 5333 digital voltmeter which is interfaced to the PET microcomputer. A multiplexer permits use of the DVM in measuring several

Figure 15. A block diagram of the electronics of the a.c. calorimeter.



different voltages in the experiment. The interface between the PET and voltmeter is the ICS Electronics Corporation Model 4880 Instrument Coupler which can be connected between an IEEE 488 port and any digital instrument and can either transfer information to the PET from a measuring device or commands from the PET to the various devices. At each temperature 60-200 readings of the DVM output, taken at 1 second intervals, are averaged and that average and its standard deviation are recorded on tape by the PET.

The vacuum can and heat sink each have heater wires wound on them and are covered with Sauereisen cement.<sup>33/</sup> The vacuum can's temperature is controlled by a d.c. bridge.<sup>20/</sup> The temperature control circuit for the heat sink is shown in Figure 16, where one leg of the a.c. bridge is a General Radio Model 1435 Programmable Decade Resistor which is interfaced to the PET. The temperature of the heat sink is changed by taking discrete steps in resistance which correspond to temperature jumps of 1 to 20 mK. At each step the output of the a.c. bridge is monitored by a P.A.R. HR8 Lock-in amplifier connected to the DANA 5333 which is interfaced with the PET. The PET steps the resistance and then waits for the HR8 output to reach a null,  $V \leq 1$  volt, before taking the experimental data. The temperature of the bath is measured with a platinum thermometer<sup>34/</sup> hooked to a constant current source and to the DVM. Both before and after the averaging of the 128 signal the voltage across the platinum thermometer is measured with two directions of current, averaged, and recorded by the PET. In Figure 17 a flow chart of the BASIC program used to control the experiment and take the data shows the order of the various control steps and measurements for each data point.



Figure 16. The a.c. bridge used to control the temperature of the heat sink.

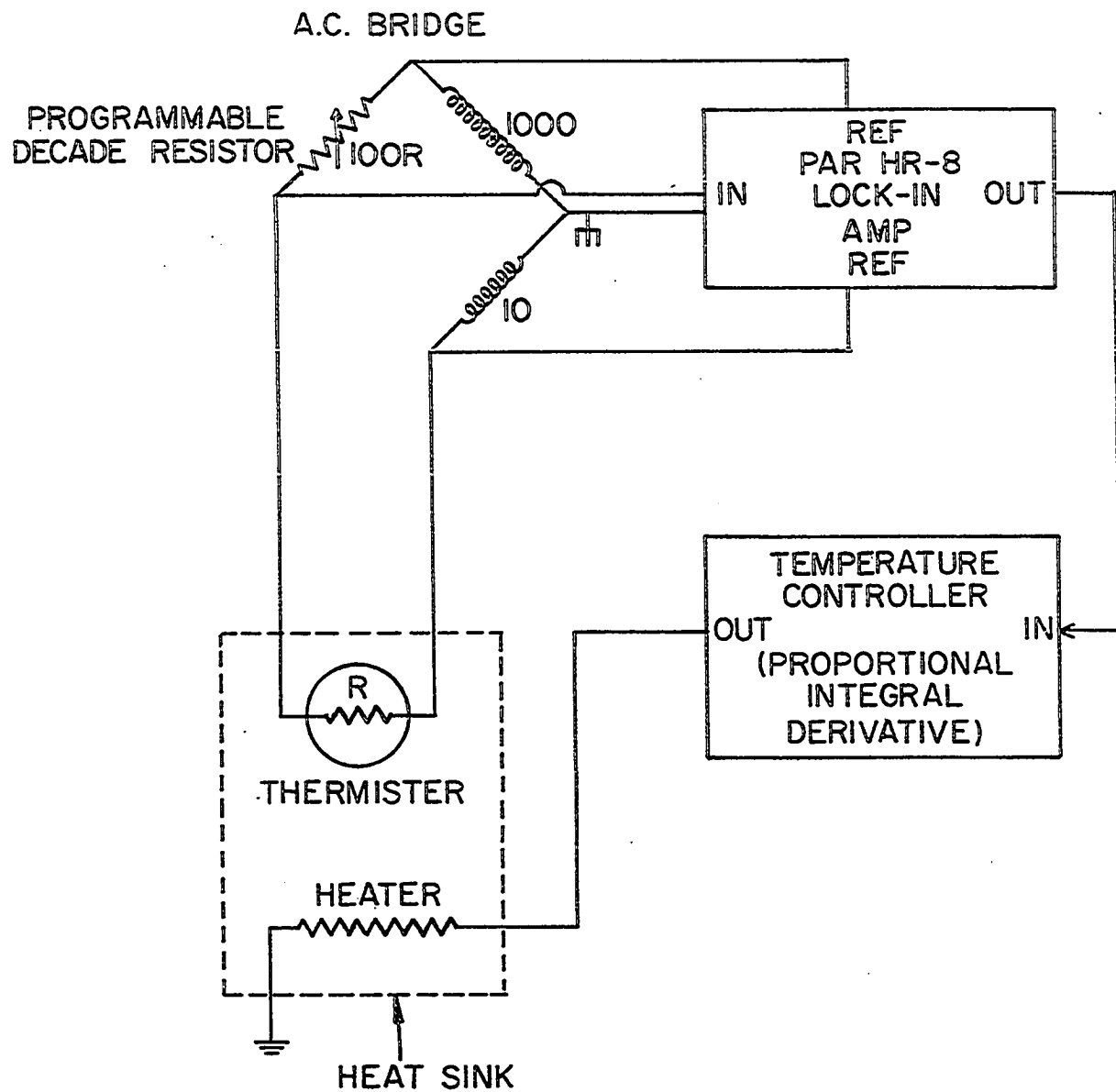
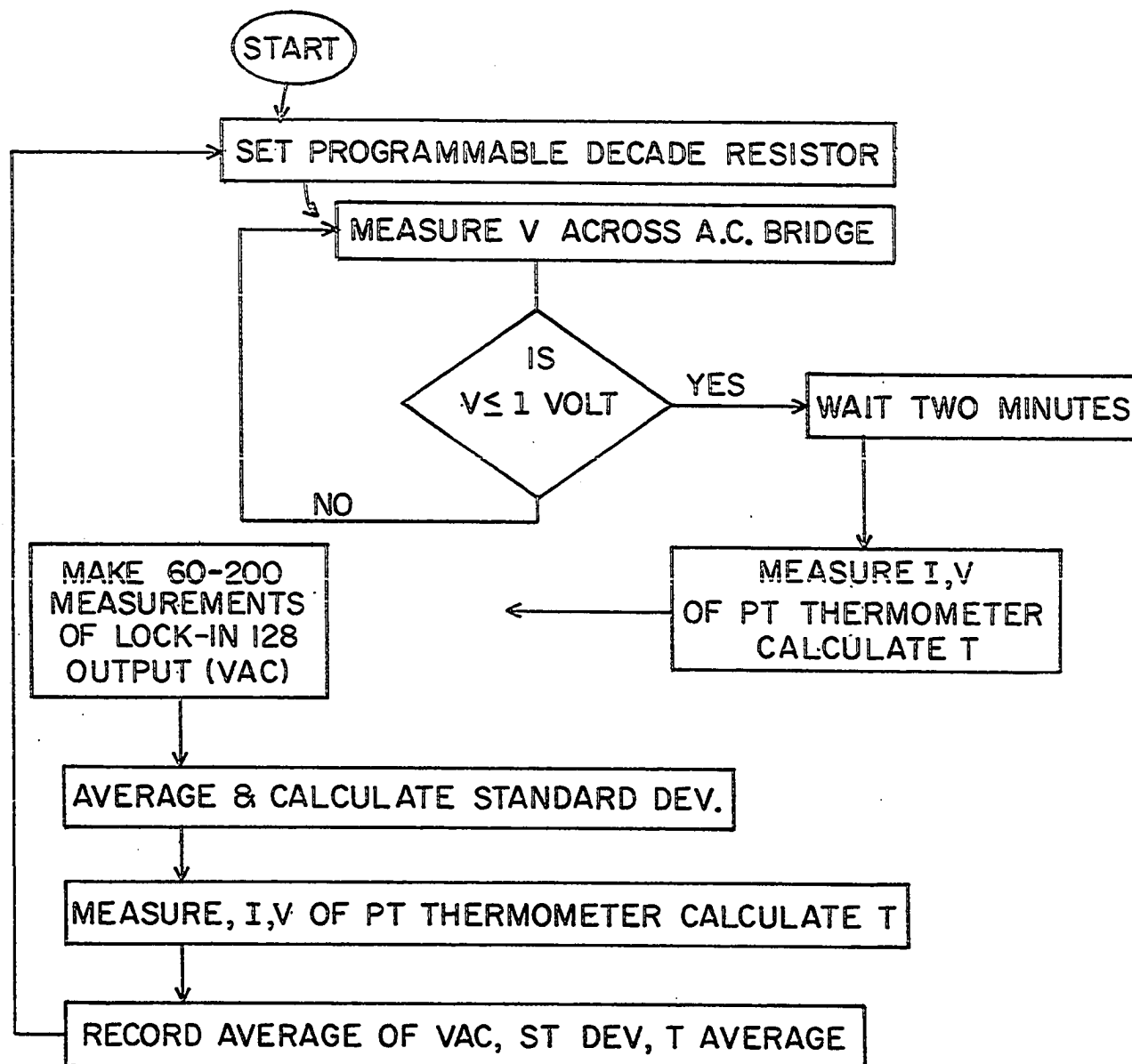


Figure 17. The flow chart of the BASIC program to control the experiment and record the data.

## BASIC FLOW CHART



### 3.5 The Data Analysis

The heat capacity data is fit to the following equations:

$$C_p^+ = \frac{A^+}{\alpha} \left[ |t|^{-\alpha} - 1 \right] + B^+ + Dt \quad (3.7)$$

$$C_p^- = \frac{A^-}{\alpha} \left[ |t|^{-\alpha} - 1 \right] + B^- + Dt \quad , \quad (3.8)$$

where  $t = (T - T_c)/T_c$  is reduced temperature.  $+$  refers to temperature above  $T_c$  and  $-$  to temperature below  $T_c$ . The fitting procedure assumes that both  $T_c$  and  $\alpha$  are the same on both sides of the transition as expected for a critical point. The programs used are based on programs of Salamon and Lederman.<sup>35/</sup> The data below  $T_c$  are interpolated so that for every point there is a point above  $T_c$  at the same absolute value of reduced temperature. The program then subtracts the linear background from the data points and does a linear least squares fit for  $C_p^+$  as a function of  $C_p^-$ . The linear relationship between  $C_p^+$  and  $C_p^-$  is based on the assumption that the exponent  $\alpha$  is the same on both sides of the transition.

$$C_p^+ = \left( \frac{A^+}{A^-} \right) C_p^- + B^+ - (A^+ B^- / A^-) \quad . \quad (3.9)$$

The critical amplitude ratio  $A^+/A^-$ , which is the slope of the line, and  $B^+ - (B^- \times A^+/A^-)$ , which is the intercept, are determined by the linear fit. Different values for the background slope,  $D$ ,  $T_c$ , and the range of data fit are tried until the linear fit is optimized. The program calculates  $\chi^2$  for the linear fit and a linear correlation coefficient, which should approach 1. If the phase transition is not second order the assumption  $\alpha^+ = \alpha^-$  will not hold, and a good linear fit cannot be obtained. The linear relationship between  $C_p^+$  and  $C_p^-$  is used to merge the  $C_p^-$  data with  $C_p^+$

data. A three parameter non-linear least squares fit, based on the technique called Marquadt's compromise,<sup>36/</sup> is done on this merged set of data to find values for  $\alpha$ ,  $A^+$ , and  $B^+$ .  $\chi^2$  is calculated using the standard deviations measured for each data point.  $A^-$  and  $B^-$  are calculated from the slope and intercept resulting from the linear fit. Although it is not assumed that  $B^+ = B^-$  it has come out of the fitting procedure that they are the same, as theoretically predicted.<sup>37/</sup>

The data analysis is done for different ranges of reduced temperature to find over what range the fitted parameters are not sensitive to range shrinking. Plots of  $\log C'$  vs  $\log t$ , where

$$C'^+(t) = \left[ C_{\text{data}}^+(t) - (B^+ + Dt - (A^+/\alpha)) \right] \alpha/A^+ \quad \text{and} \quad (3.10)$$

$$C'^-(t) = \left[ C_{\text{data}}^-(t) - (B^- + Dt - (A^-/\alpha)) \right] \alpha/A^-, \quad (3.11)$$

show where the experimental rounding occurs and indicate over what range the heat capacity obeys a power law. If the rounded region near  $T_c$  is included the value of  $\alpha$  decreases. In the correct range of reduced temperature the parameters resulting from the fit are not sensitive to range shrinking.

The uncertainties in the exponent alpha are obtained by finding a 95% confidence range using the F-Test. For data in a given range of reduced temperature a plot is made of chi squared as a function of  $\alpha$ , where each  $\alpha$  is part of a parameter set including  $A^+$  and  $B^+$ , and the sets of parameters accepted are those for which

$$\left( \chi^2 - \chi_o^2 \right) / \chi_o^2 \leq \frac{p}{n-p} \left[ F_{o,.05}(p, n-p) \right] \quad (3.12)$$

where  $p=3$ , the number of parameters fit,  $n$  = the number of data points,  $\chi_0^2$  is the minimum value of chi squared, and  $F_{0,.05}$  is the 5% F distribution parameter. There is a 95% probability that the true value of  $A^+$ ,  $\alpha$ , and  $B^+$  lie within the parameter space defined by the above equation.<sup>38/</sup>

## CHAPTER 4

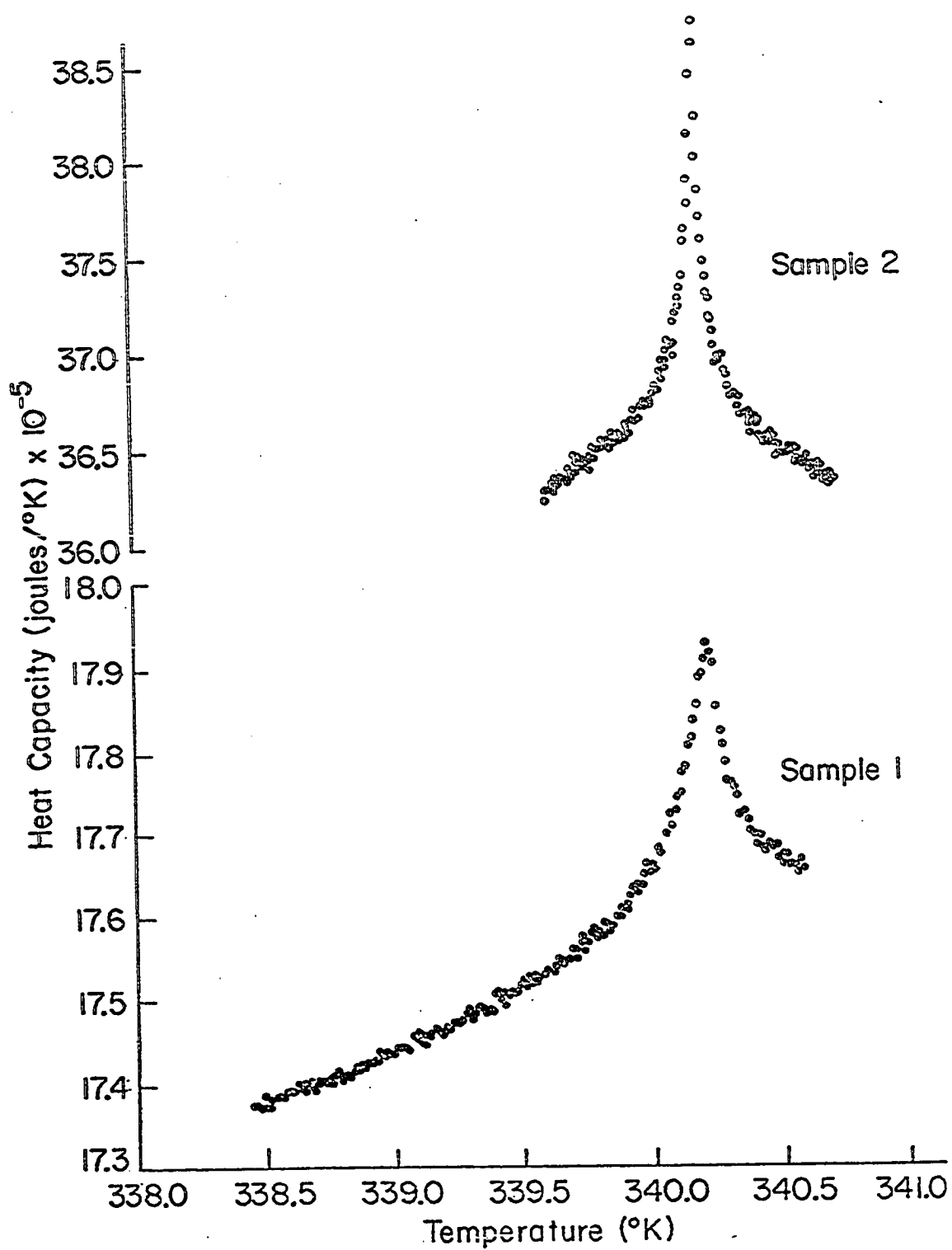
## RESULTS AND DISCUSSION

4.1 Octyloxycyanobiphenyl Data and Chemical Analysis

The heat capacity data for the two samples of 80CB appear in Figure 18 where sample 1 is the earlier sample vacuum sublimated by Cladis. Two independent runs have been done on this sample, one at 1.96 Hz in August and the other at 9 Hz in December. The runs produce similar results for the background slope,  $\alpha$ , and the critical amplitude ratio. The data for sample 1 in Figure 18 is from the December run on this sample. The rms value of the temperature oscillation is 5 mK. The heater power is on the order of  $10^{-4}$  watts. Sample 2 in Figure 2 is the later sample. The rms temperature oscillation is 1-2 mK enabling us to fit the data closer to  $T_c$ . The frequency of the heater voltage is again 9 Hz and the power is on the order of  $10^{-4}$  watts. The difference in the  $T_c$ s of the two samples is  $32 \text{ mK} \pm 35 \text{ mK}$ . This is the difference between the average  $T_c$  from the two runs on sample 1 and the  $T_c$  of sample 2. The transition temperatures are stable through repetitions of the experiment on each sample. The transition appears to be second order with a sharp symmetric peak for both samples within the experimental resolution. Neither flattening of the peak nor a two phase region is observed in either sample. No hysteresis between heating and cooling sweeps is observed. The transition may still be slightly first order but even in this case critical behavior is observed close to  $T_c$ .



Figure 18. The data on sample 1 is from the December run. The shapes of these curves illustrate the different behavior of these samples near the nematic-smectic A transition. In spite of the agreement between transition temperatures sample 1 produces  $\alpha \approx 0$  and sample 2's results show  $\alpha \approx .25$ .



The values of alpha and other fitted parameters are listed in Table 2. More detailed tables of the results of fits on all the runs for different ranges of reduced temperature on the different samples appear in the Appendix. If data close to  $T_c$  are included, the value of alpha obtained from the fit is smaller due to experimental rounding. The range of data that should be fitted includes those points where the resulting fitted parameters are not sensitive to range shrinking. The log-log plots for the two samples of 80CB, shown in Figure 19, indicate the range of data where the heat capacity follows a power law and is not affected by rounding. The values shown in Table 2 are the results of fitting over a temperature range not affected by rounding and are those values which produce the best fits.

TABLE 2  
SUMMARY OF RESULTS FOR SAMPLES OF 80CB

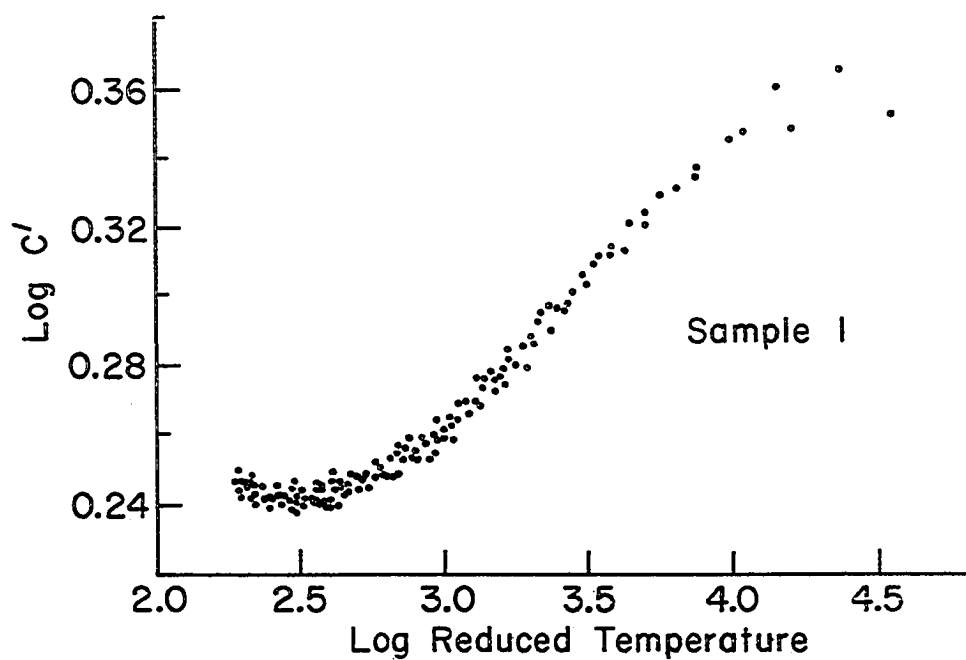
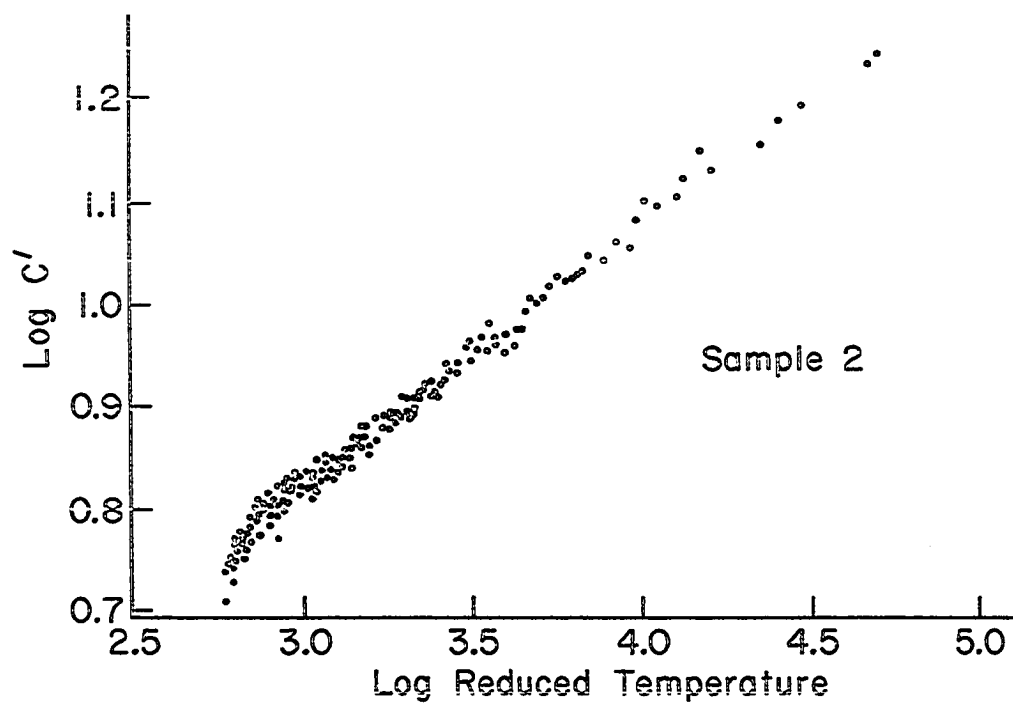
Sample	Range of $-\log t $	$T_{dc}$ (mK)	$T_{ac}$ (mK)	$T_c$ (K)	$\alpha$	$A^+/A^-$
1 - 8/79 run	2.3-4.2	154	10	$340.107 \pm 3^a$ mK, $\pm 40^b$ mK	$.025 \pm .04$	1.39
1 - 12/79 "	2.9-4.0	---	5	$340.215 \pm 7^a$ mK, $\pm 50^b$ mK	$.034 \pm .03$	1.3
2	2.9-4.6	100	1-2	$340.193 \pm 2^a$ mK, $\pm 15^b$ mK	$.25 \pm .02$	1.2

<sup>a</sup> These errors are due to uncertainty in fitted value of  $T_c$ .

<sup>b</sup> These errors are due to uncertainty in dc temperature use.

The two different samples of 80CB measured are equivalent to those samples measured by Johnson<sup>15/</sup> and Garland.<sup>16/</sup> Johnson finds  $\alpha \leq$  zero in agreement with deGennes<sup>13/</sup> superconducting analogy and  $A^+/A^- = 1.35$

Figure 19. Log-log plots for the two samples of 80CB. The data for sample 1 where  $\alpha \approx 0$  rounds over further from  $T_c$  than the data for sample 2 where  $\alpha \approx .25$ . Such plots indicate over which range of temperature the data can be analyzed without rounding affecting the exponent.



in a.c. heat capacity measurements on an early sample of 80CB. This agrees with the result for sample 1, the earlier sample shown in Table 2 ( $\alpha = .03 \pm .03$ ,  $A^+/A^- = 1.35$ ). Garland's measurements on a later batch of 80CB show  $\alpha = .25 \pm .05$  and  $A^+/A^- = 1.0$ . Sample 2 is from that same batch and yields  $\alpha = .25 \pm .02$  and  $A^+/A^- = 1.2$ . The data from sample 1 cannot be fit to a power law with  $\alpha = .25$  in any range of reduced temperature. Thus the lower exponent of sample 1 is not an artifact of experimental rounding due to impurities, but is indicative of different critical behavior.

The fact that the transition temperatures are in agreement is evidence that the two samples are the same compound. High pressure liquid chromatography<sup>39/</sup> has shown that each sample has three overlapping peaks with a slight difference in the height of one of the peaks. This indicates the presence of impurities similar to the main compound; such as positional isomers. Homologous impurities may be ruled out as pure octylbromide would be used in the alkylation of the molecule. 220 MHz NMR has indicated the presence of only one component in each sample, although impurity levels on the order of 1% are difficult to detect with NMR. British Drug Houses<sup>40/</sup> reports a purity of greater than 99.9% for the later batch of 80CB where .01% impurities have been detected by gas liquid chromatography. The comparison of transition temperatures and chromatography data indicate that the samples are similar in composition. The information from British Drug Houses and the NMR data show that the level of impurities in each sample is less than a percent. The conclusion then is that the two samples of 80CB have small but differing concentrations of impurities which might be isomeric impurities.

#### 4.2 Evidence for a Crossover Point

The fact that the two samples of 80CB behave so differently near the transition suggests that 80CB may be near a crossover point, such that a small change in composition places it in a different region of the phase diagram. Lubensky and Chen propose a crossover from anisotropic behavior to helium-like behavior followed by a tricritical point. Anisotropic behavior is characterized by  $v_{||}$  and  $v_{\perp}$  diverging at different rates as observed by Birgeneau<sup>17,18/</sup> Lubensky and Chen<sup>1/</sup> predict that when  $K_1$ , the splay elastic constant, is very large, anisotropy results because splay is forbidden and the smectic layers cannot bend. As  $T_c$  is approached  $K_2$  and  $K_3$  are enhanced and when they are on the order of  $K_1$  there is a crossover to isotropic helium-like behavior.

A critical point or second order transition is a fixed point unstable with respect to one variable such as temperature. A crossover point, however, is a fixed point which is unstable with respect to two variables. Exponents must be measured with respect to temperature and one other parameter to study a possible crossover.

The phase diagram for the nematic-smectic A transition may be studied by measuring  $\alpha$  for several compounds where each compound has a different value of the  $T_{na}/T_{ni}$  ratio. Table 3 shows results for all of the compounds studied including the value of  $T_{na}/T_{ni}$  for each material. Detailed tables of results for different temperature ranges on different sets of data are shown in the Appendix for all of the samples. All of the measurements have been taken with temperature oscillations on the order of 1-5 mK. CBOOA and CBNOA were measured at  $f=9$  Hz, 8CB at  $f=1.4$  Hz, and 8S5 at 3.17 Hz. The results indicate two types of critical behavior with

TABLE 3

## SUMMARY OF RESULTS

Sample	T <sub>na</sub> /T <sub>ni</sub>	T <sub>ac</sub> (mK)	T <sub>c</sub> (K)	T <sub>dc</sub> (mK)	-log t  Range	$\alpha$	A <sup>+</sup> /A <sup>-</sup>
CBNOA	.99	2-3	Heating 370.1 $\pm$ 1.1 <sup>b</sup> Cooling 370.6 $\pm$ 1.1 <sup>b</sup>	-----	-----	-----	-----
8CB	.98	1	307.017 $\pm$ 1 <sup>a</sup> mK, $\pm$ 12 <sup>b</sup> mK	53	2.6-4.0	.285 $\pm$ .03	.971
8OCB (2)	.96	1-2	340.193 $\pm$ 2 <sup>a</sup> mK, $\pm$ 15 <sup>b</sup> mK	100	2.9-4.6	.25 $\pm$ .02	1.2
8OCB (1-8/79)	.96	10	340.107 $\pm$ 3 <sup>a</sup> mK, $\pm$ 40 <sup>b</sup> mK	154	2.3-4.2	.025 $\pm$ .04	1.39
8OCB (1-12/79)	.96	5	340.215 $\pm$ 7 <sup>a</sup> mK, $\pm$ 50 <sup>b</sup> mK	-----	2.9-4.0	.034 $\pm$ .03	1.3
CB00A	.94	3-4	356.950 $\pm$ 1 <sup>a</sup> mK	713	2.2-3.6	-.12 $\pm$ .02	.778
8S5	.94	2-3	336.486 $\pm$ 1 <sup>a</sup> mK, $\pm$ 50 <sup>b</sup> mK	$\lesssim$ 100	2.5-4.0	-.17 $\pm$ .03	.575

<sup>a</sup>Errors due to uncertainty in data fitting.<sup>b</sup>Errors due to uncertainty in calculating dc temperature rise.



a crossover point at  $T_{na}/T_{ni}=.96$ , the value corresponding to 80CB. Those compounds, whose ratio is below this value, have values of  $\alpha$  less than or equal to zero. More rounding is observed around  $T_c$  so the data can not be fit as closely to  $T_c$  as with those runs where  $T_{na}/T_{ni} \geq .96$ . The linear fit is described by linear correlation coefficients which are typically around .98. The data for  $\overline{8S5}$  and CB00A appear in Figures 20 and 21. 8CB, whose  $T_{na}/T_{ni}$  ratio is greater than .96, produces values of  $\alpha$  around .3 and has a very sharp peak. The data, which is shown in Figure 22, can be analyzed to within several mK of  $T_c$  and the linear fits are characterized by linear correlation coefficients between .99 and 1.0. Clearly there are two types of critical behavior exhibited as a function of  $T_{na}/T_{ni}$ , where the crossover occurs at  $T_{na}/T_{ni}$  equal to .96.

CBNOA has  $T_{na}/T_{ni}$  equal to .99 and exhibits a first order transition as evidenced by the  $.5^\circ$  shift between the transition temperatures of the heating and cooling sweeps. This is illustrated in the data in Figure 23. There is also no pretransitional behavior observed, as there are no fluctuations characteristic of a critical point in CBNOA.

Summarizing these results, there are three regions of the phase diagram. Sample 1 of 80CB,  $\overline{8S5}$ , and CB00A have rounded regions close to  $T_c$  and values of  $\alpha$  near zero or negative. Sample 2 of 80CB and 8CB have sharp symmetric peaks with a large positive value of .3 for  $\alpha$ . The crossover between these two types of critical behavior occurs at  $T_{na}/T_{ni}=.96$ . CBNOA exhibits strong first order behavior. It appears that there is a tricritical point between  $T_{na}/T_{ni}=.98$ , which is the value for 8CB, and  $T_{na}/T_{ni}=.99$ , the value for CBNOA.

Figure 20. Heat capacity data for  $\overline{8S5}$  where  $\alpha$  equals  $-.17 \pm .03$  and  $A^+/A^- = .575$ . Rounding appears at  $-\log|t| = 3.6$ . This is possibly a first order transition as the log-log plot shows  $\alpha^+ \neq \alpha^-$  at  $-\log|t| \approx 3.6$ .

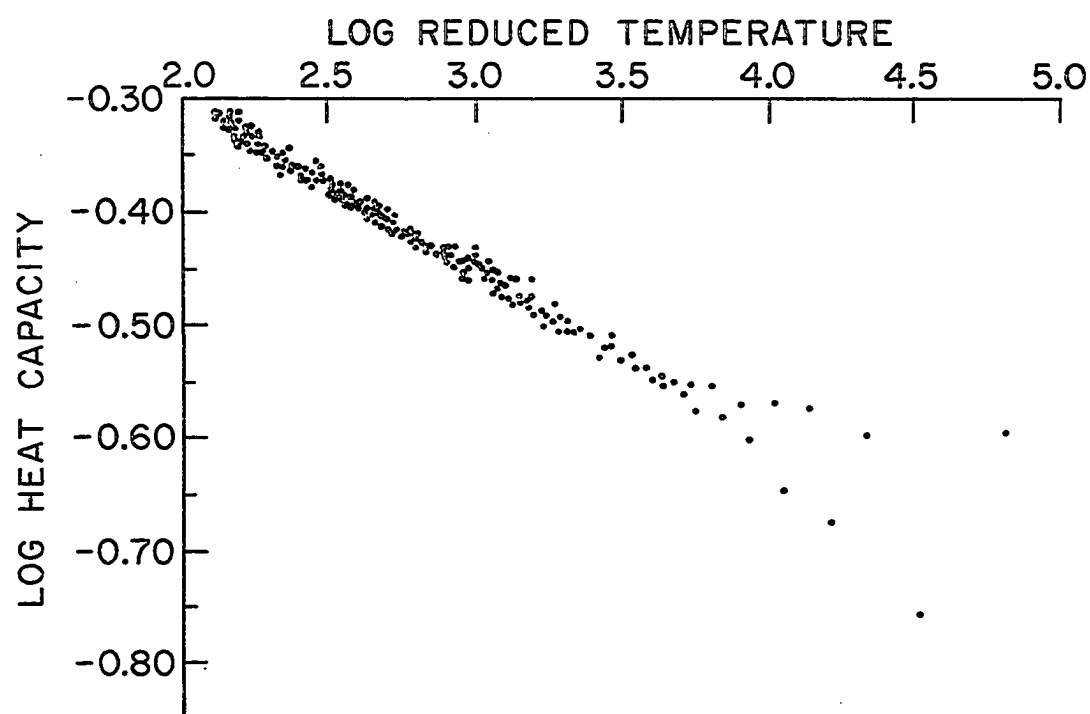
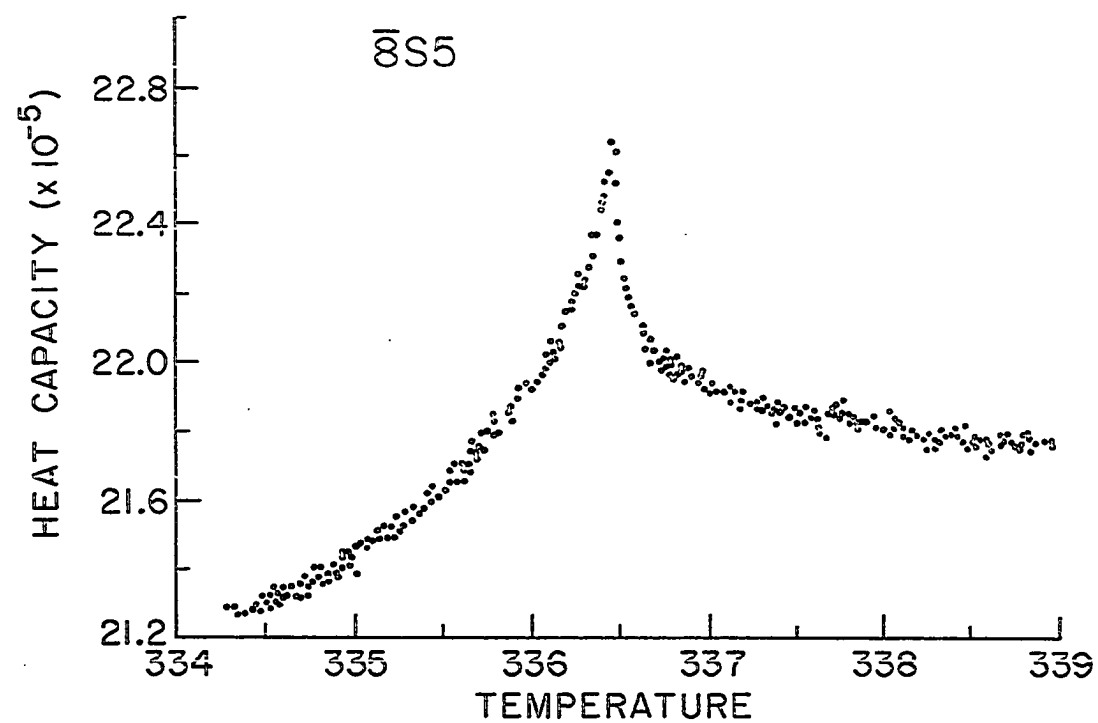


Figure 21. Heat capacity data for CB00A where  $\alpha = -.12 \pm .02$  and  $A^+/A^- = .778$ . Rounding occurs at  $-\log|t| = 3.7$ .

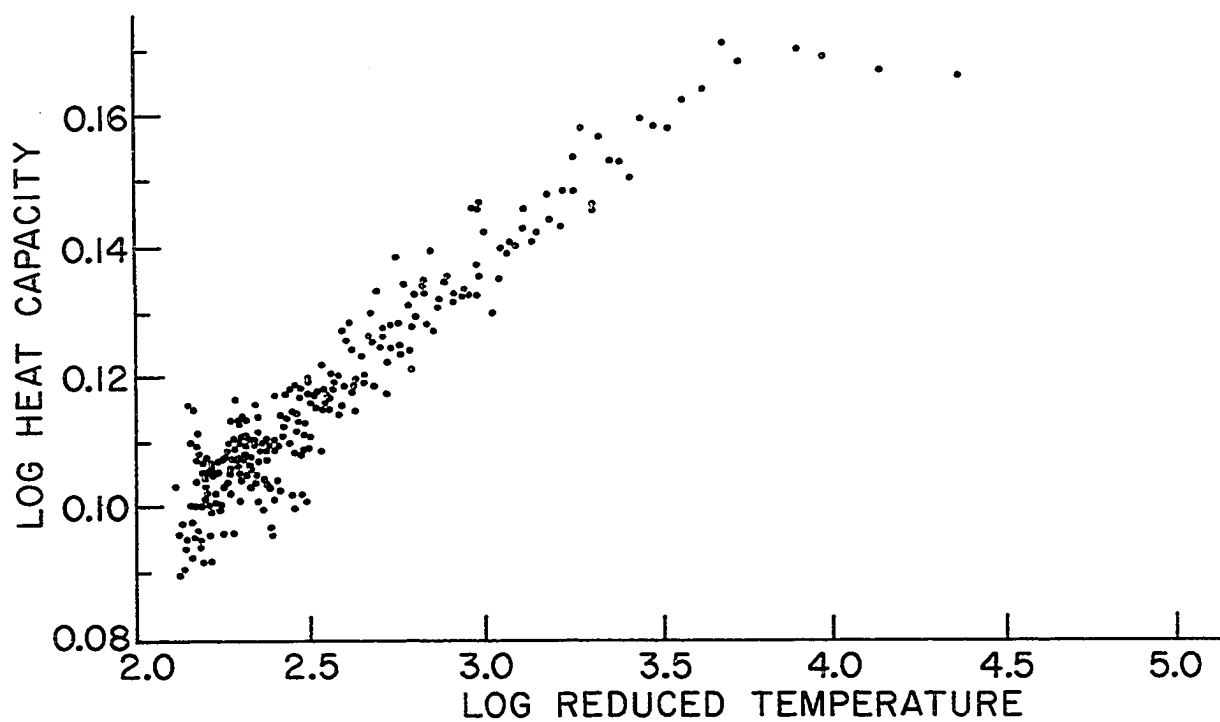
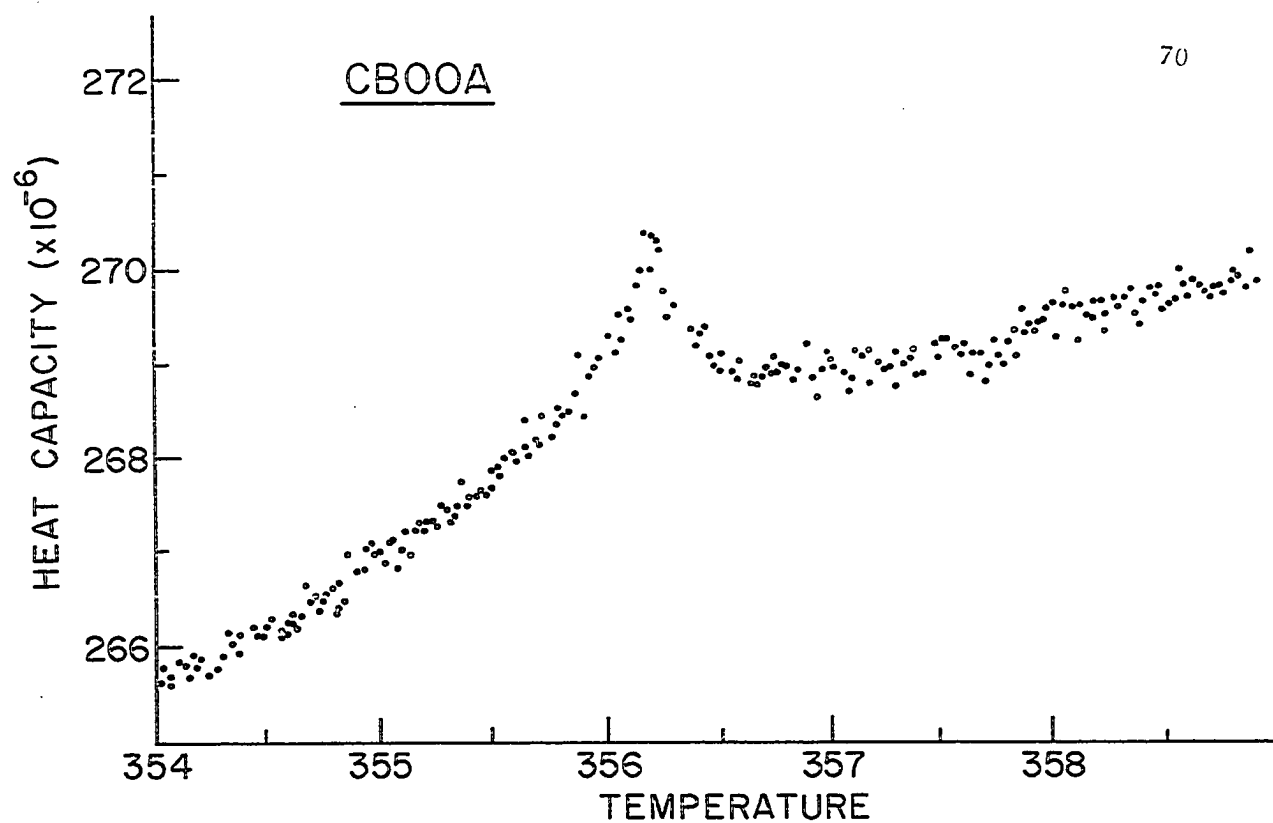


Figure 22. Heat capacity data for 8CB. The exponent  $\alpha$  is  $.285 \pm .03$  and  $A^+/A^- = .971$ . Rounding appears at  $-\log|t| = 4.5$ .

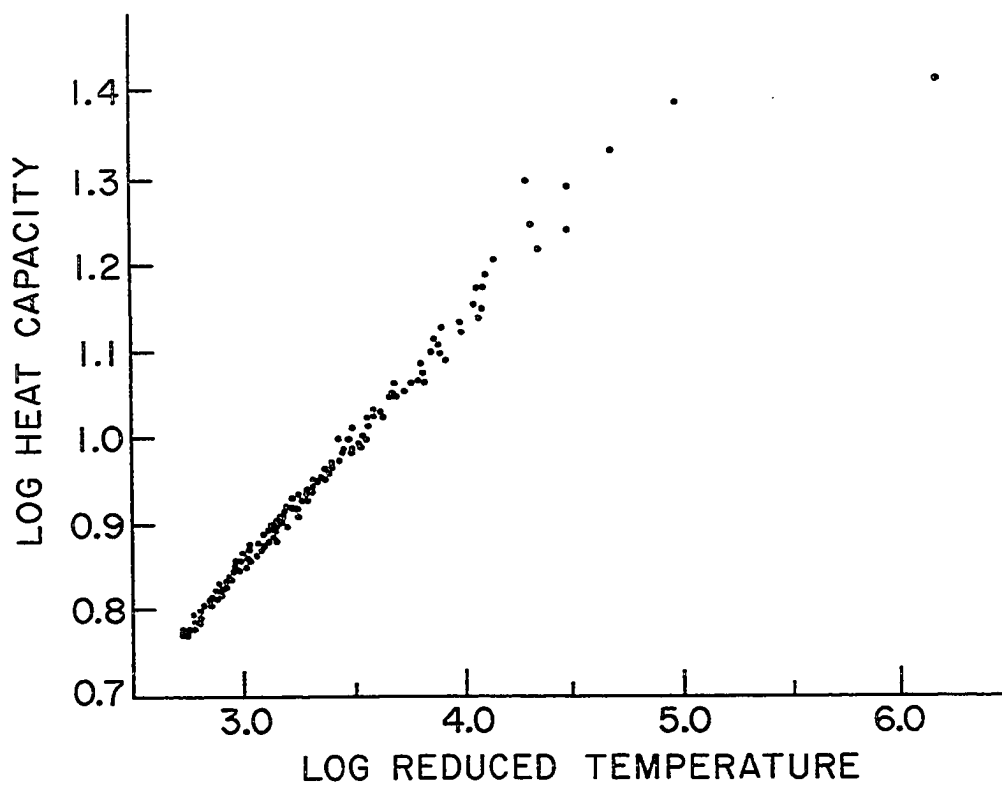
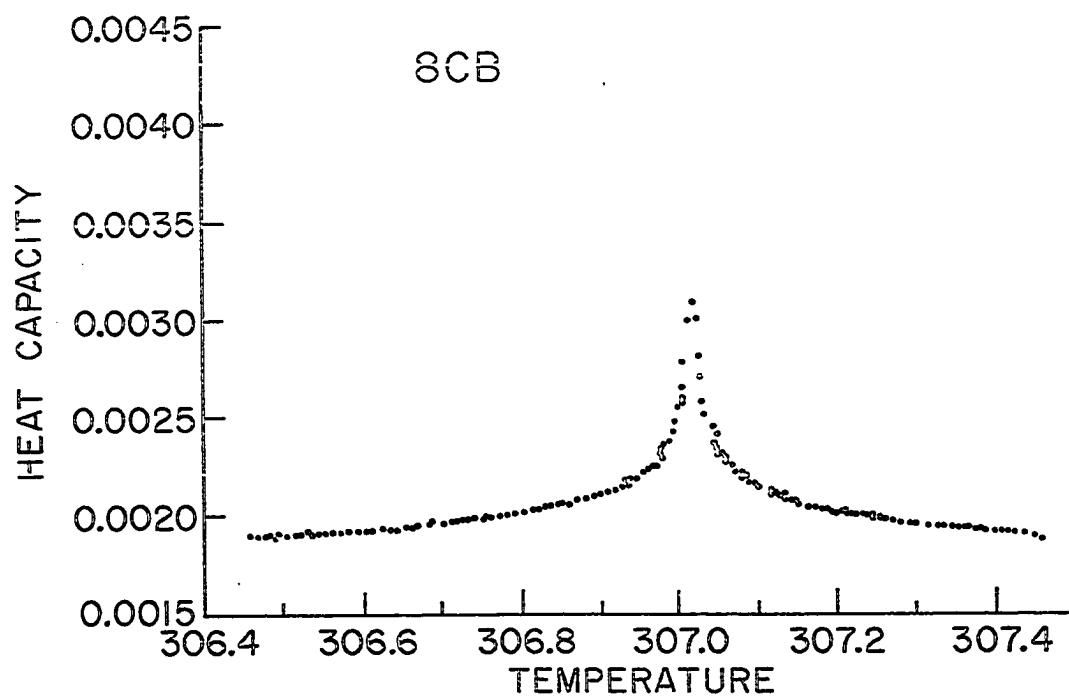
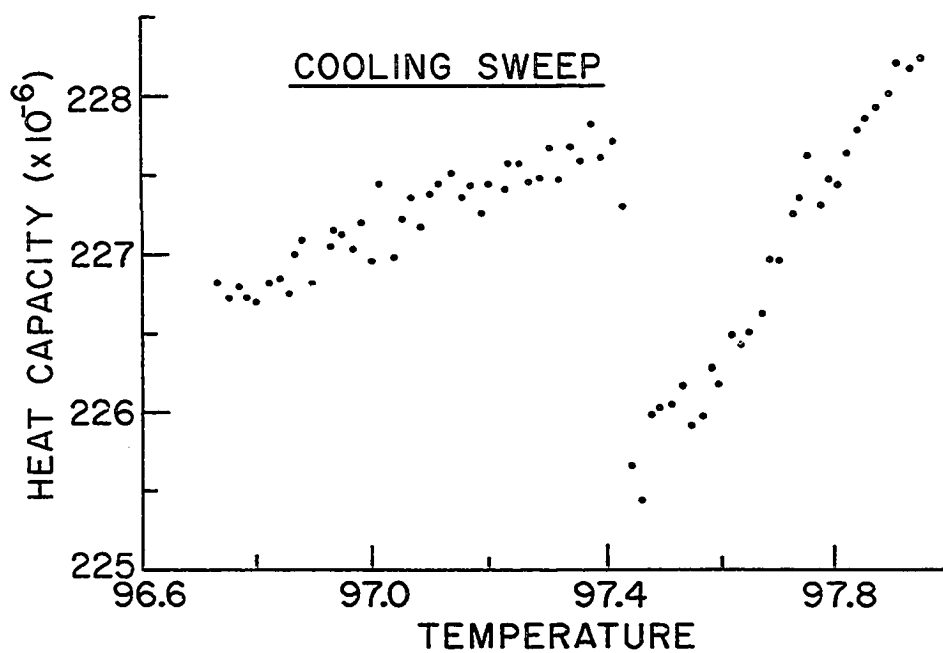
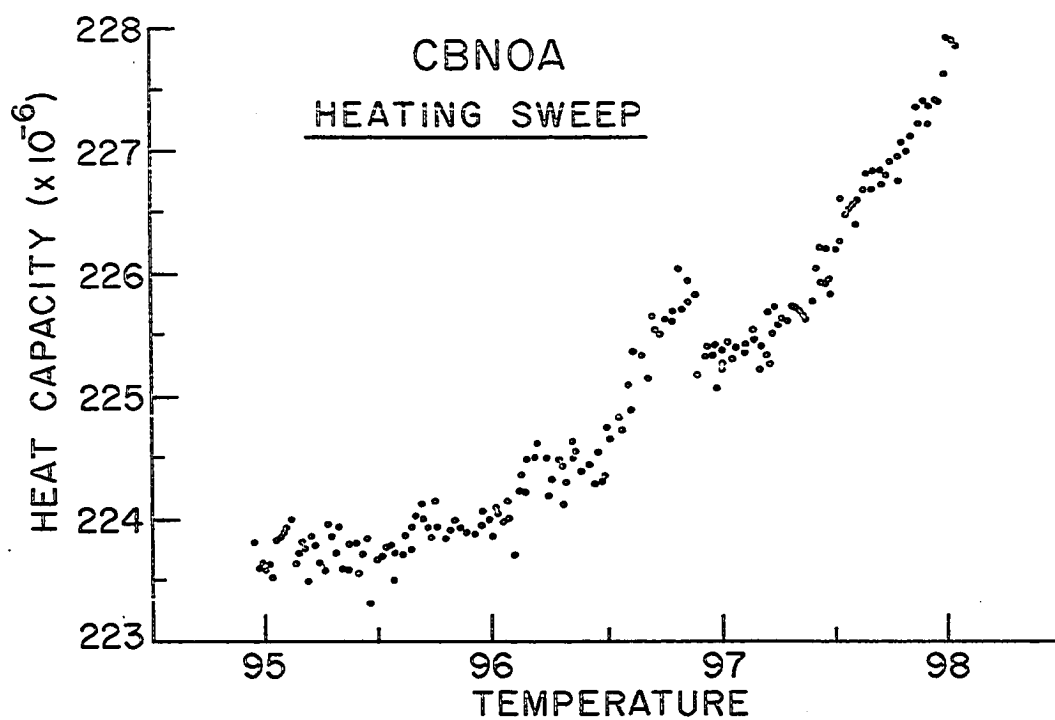


Figure 23. Heat capacity data for CBNOA showing the  $.5^{\circ}$  shift in  $T_c$  between the heating and cooling sweeps. This hysteresis indicates a strong first order transition.





Garland<sup>16/</sup> has done measurements of heat capacity on 80CB at high pressures where increased pressure produces a smaller value of  $T_{na}/T_{ni}$ . At 500 bar,  $T_{na}=347.76$ ,  $T_{ni}=369.39$ , and  $T_{na}/T_{ni}=.94$  as compared with .96 at 1 bar. Garland observes a decrease in the amplitude of the peak at 500 bar and again at 1000 bar, but the exponent equals .25 at all pressures. This disagrees with the results in Table 3 which show lower exponents for CBOOA and  $\overline{8S5}$ , whose ratios are also .94. It also disagrees with Birgeneau's result for CBOOA that  $\alpha$  equals .06 when  $v_{||}$  and  $v_{\perp}$  are substituted into the Josephson scaling law. A comparison with the  $\alpha$ s calculated from Birgeneau's measurements, however, may not be valid as the error bars on the  $\alpha$  calculated from his x-ray results are large.

Brisbin et al.<sup>41/</sup> have done heat capacity and birefringence measurements on three members of the  $\overline{nS5}$  series and have obtained an indication of the crossover behavior exhibited by the data in Table 3. These results appear in Table 4, where the values of  $T_{na}/T_{ni}$  are in the range .94-.99. Helium-like behavior is exhibited at  $T_{na}/T_{ni}=.94$ . At  $T_{na}/T_{ni}=.966$  a crossover has occurred to quasicritical behavior with  $\alpha = .25$ . At larger values of  $T_{na}/T_{ni}$  there is a tricritical point. This is indicated by the tricritical value of .5 for  $\alpha$  at  $T_{na}/T_{ni}=.986$  and the increasing hysteresis ( $T_C - T_C'$ ). This sequence of behavior qualitatively agrees with the results in Table 3 and agrees with the prediction made by Lubensky and Chen<sup>1/</sup> that there would be a crossover between two types of critical behavior as well as a tricritical point. The sequence of behavior predicted by Lubensky and Chen, anisotropic to helium-like to first order, however, does not agree with the experimental results.

TABLE 4  
SUMMARY OF RESULTS OF BRISBIN ET AL.<sup>41/</sup>

Compound	T <sub>na</sub> /T <sub>ni</sub>	$\alpha$	T <sub>c</sub> - T <sub>c</sub> '
$\overline{8S5}$	.94	0 $\pm$ .02	0 $\pm$ .007
$\overline{9S5}$	.97	.22 $\pm$ .03	.002 $\pm$ .01
$\overline{10S5}$	.986	.45 $\pm$ .05	.02 $\pm$ .01

#### 4.3 Future Directions

The measurement of exponents would be more meaningful if better work were being done to determine whether the nematic-smectic A transition is first or second order. It has still not been resolved if this transition is always first order as predicted by Halperin, Lubensky, and Ma.<sup>42/</sup> Although the assumption that  $\alpha^+ = \alpha^-$  is valid even in slightly first order transitions, it would be interesting to follow the change in critical behavior as a function of the latent heat or of the size of the discontinuity in the order parameter. It is not clear whether the transitions observed in the compounds in Table 3 are purely second or slightly first order. Knowledge of this might explain why there is more rounding when the exponent is smaller. Brisbin et al.<sup>41/</sup> incorporate birefringence measurements with high resolution specific heat measurements on the  $\overline{nS5}$  series and observe a crossover point and a tricritical point. Similar measurements should be done on other series of compounds of exponents and of the order of the phase transition.

A complete physical picture of the nematic-smectic A transition will not emerge until there is agreement among different research groups on the

exponents and order of the transition for many compounds. Such repetition is necessary because of the difficulty of obtaining critical exponents on liquid crystals and the problems with chemical characterization. Better chemical analysis and purification techniques are important for more reliable measurements.

Another interesting possibility is to dope a pure liquid crystal, which is near the crossover point, such as 80CB, with controlled amounts of known impurity such as isomers or a neighboring homolog and attempt to shift the transition to the other side of the crossover point. The physical understanding of how a given impurity affects the ordering of the liquid crystal and its critical behavior would be important to the study of phase transitions in other systems. The region around the crossover point might also be explored by studying mixtures of two homologs and observing the exponent  $\alpha$  as a function of concentration of one homolog in another. This would give a more detailed picture of the phase diagram around the crossover or tricritical point.

The motivation to understand the nematic-smectic A transition is that once it is understood an experimentally accessible system will exist where the effect of impurities on critical behavior and special crossover points can be studied in detail. The research presented in this thesis indicates that this transition may exhibit both a crossover point between two critical behaviors as well as a tricritical point and, therefore, may prove to be important to phase transition physics.

## APPENDIX

This appendix contains detailed tables of the results of the fitting procedure on the different experiments on the different samples. The best fits are summarized in Chapter 4 and are those fits which are not affected by rounding and which fit the data without systematic deviations.

The error bars on  $T_c$  are expressed in two parts. The smaller uncertainty,  $a$ , is the uncertainty due to the fit while  $b$  is the uncertainty in  $T_c$  due to the uncertainty in the d.c. temperature rise.

80CB SAMPLE 1 (8/79)

f (Hz)	T <sub>ac</sub> (mK)	T <sub>dc</sub> (K)	T <sub>c</sub> (K)	D (J/K)	-log t	χ <sup>2</sup>	A <sup>+</sup> /A <sup>-</sup>	A <sup>+</sup> (J/K)	B <sup>+</sup> (J/K)	A <sup>-</sup> (J/K)	B <sup>-</sup> (J/K)	α
1.96	10	.154	340.107 ±3 <sup>a</sup> <sub>mk</sub> , ±40 <sup>b</sup> <sub>mk</sub>	.45×10 <sup>-3</sup>	2.9-4.2	.797	1.36	.17 ×10 <sup>-5</sup>	.167×10 <sup>-3</sup>	.13 ×10 <sup>-5</sup>	.171×10 <sup>-3</sup>	-.026
1.96	10	.154		.45×10 <sup>-3</sup>		.810		.88 ×10 <sup>-6</sup>	.17 ×10 <sup>-3</sup>	.65 ×10 <sup>-6</sup>	.179×10 <sup>-3</sup>	.058
1.96	10	.154		.45×10 <sup>-3</sup>		.807		.94 ×10 <sup>-6</sup>	.17 ×10 <sup>-3</sup>	.69 ×10 <sup>-6</sup>	"	.05
1.96	10	.154		.45×10 <sup>-3</sup>		.948		.5 ×10 <sup>-5</sup>	.157×10 <sup>-3</sup>	.37 ×10 <sup>-5</sup>	.164×10 <sup>-3</sup>	.165
1.96	10	.154		.45×10 <sup>-3</sup>		.924		.387×10 <sup>-6</sup>	.193×10 <sup>-3</sup>	.285×10 <sup>-6</sup>	.176×10 <sup>-3</sup>	+.16
1.96	10	.154		.45×10 <sup>-3</sup>		.833		.26 ×10 <sup>-5</sup>	.164×10 <sup>-3</sup>	.191×10 <sup>-5</sup>	.169×10 <sup>-3</sup>	-.08
1.96	10	.154		.53×10 <sup>-3</sup>	2.3-5.0	1.82	1.36	.177×10 <sup>-5</sup>	.168×10 <sup>-3</sup>	.130×10 <sup>-5</sup>	.171×10 <sup>-3</sup>	-.05
						1.46		.173×10 <sup>-5</sup>	.168×10 <sup>-3</sup>	.127×10 <sup>-5</sup>		-.05
						1.43		.17 ×10 <sup>-5</sup>	.168×10 <sup>-3</sup>	.124×10 <sup>-5</sup>		-.054
						1.70		.146×10 <sup>-5</sup>	.169×10 <sup>-3</sup>	.107×10 <sup>-5</sup>	.172×10 <sup>-3</sup>	-.02
1.96	10	.154	340.107 ±3 <sup>a</sup> <sub>mk</sub> , ±40 <sup>b</sup> <sub>mk</sub>	.35×10 <sup>-3</sup>	2.9-5.0	1.5	1.25	.39 ×10 <sup>-5</sup>	.161×10 <sup>-3</sup>	.312×10 <sup>-5</sup>	.165×10 <sup>-3</sup>	-.15
						3.8		.565×10 <sup>-6</sup>	.162×10 <sup>-3</sup>	.452×10 <sup>-6</sup>	.166×10 <sup>-3</sup>	.086
						1.8		.217×10 <sup>-5</sup>	.166×10 <sup>-3</sup>	.174×10 <sup>-5</sup>	.169×10 <sup>-3</sup>	-.073
						1.5		.618×10 <sup>-5</sup>	.155×10 <sup>-3</sup>	.494×10 <sup>-5</sup>	.160×10 <sup>-3</sup>	-.2
						2.5		.302×10 <sup>-4</sup>	.107×10 <sup>-3</sup>	.242×10 <sup>-4</sup>	.122×10 <sup>-3</sup>	-.4
				.54×10 <sup>-3</sup>	2.9-4.2	1.54	1.39	.134×10 <sup>-5</sup>	.169×10 <sup>-3</sup>	.969×10 <sup>-6</sup>	.173×10 <sup>-3</sup>	-.005
						1.44		.132×10 <sup>-5</sup>	.169×10 <sup>-3</sup>	.950×10 <sup>-6</sup>	"	-.0027
						1.227		.137×10 <sup>-5</sup>	.169×10 <sup>-3</sup>	.986×10 <sup>-6</sup>	"	-.0076
						1.263		.89 ×10 <sup>-6</sup>	.17 ×10 <sup>-3</sup>	.640×10 <sup>-6</sup>	"	.054
						1.6		.399×10 <sup>-6</sup>	.172×10 <sup>-3</sup>	.287×10 <sup>-6</sup>	.175×10 <sup>-3</sup>	.169
						1.23		.12 ×10 <sup>-5</sup>	.17 ×10 <sup>-3</sup>	.863×10 <sup>-6</sup>	.173×10 <sup>-3</sup>	.009
						2.15		.8 ×10 <sup>-5</sup>	.153×10 <sup>-3</sup>	.576×10 <sup>-5</sup>	.161×10 <sup>-4</sup>	-.276
						1.25		.129×10 <sup>-5</sup>	.169×10 <sup>-3</sup>	.928×10 <sup>-6</sup>	.173×10 <sup>-3</sup>	-.0002

80CB SAMPLE 1 (12/79)

f (Hz)	T <sub>ac</sub> (mK)	T <sub>dc</sub> (K)	T <sub>c</sub> (K)	D (J/K)	-log t	χ <sup>2</sup>	A <sup>+</sup> /A <sup>-</sup>	A <sup>+</sup> (J/K)	B <sup>+</sup> (J/K)	A <sup>-</sup> (J/K)	B <sup>-</sup> (J/K)	α
8.99	5	----	340.208 ±.003 <sup>a</sup>	.42×10 <sup>-3</sup>	2.9-4.6	16.9	1.34	.117×10 <sup>-6</sup>	.174×10 <sup>-3</sup>	.823×10 <sup>-7</sup>	.175×10 <sup>-3</sup>	.271
"	"	"	"	"	"	7.3	"	.729×10 <sup>-6</sup>	.17 ×10 <sup>-3</sup>	.544×10 <sup>-6</sup>	.172×10 <sup>-3</sup>	.06
"	"	"	"	"	"	7.1	"	.8 ×10 <sup>-6</sup>	.17 ×10 <sup>-3</sup>	.597×10 <sup>-6</sup>	.172×10 <sup>-3</sup>	.045
"	"	"	"	"	"	6.8	"	.923×10 <sup>-6</sup>	.169×10 <sup>-3</sup>	.689×10 <sup>-6</sup>	.171×10 <sup>-3</sup>	.028
"	"	"	"	"	"	6.7	"	.992×10 <sup>-6</sup>	.169×10 <sup>-3</sup>	.741×10 <sup>-6</sup>	.171×10 <sup>-3</sup>	.019
"	"	"	"	"	2.9-4.0	3.79	1.314	.923×10 <sup>-6</sup>	.169×10 <sup>-3</sup>	.702×10 <sup>-6</sup>	.171×10 <sup>-3</sup>	.034
"	"	"	"	"	"	3.81	"	.937×10 <sup>-6</sup>	.169×10 <sup>-3</sup>	.713×10 <sup>-6</sup>	"	.032
"	"	----	340.223 ±.003 <sup>a</sup>	.26×10 <sup>-3</sup>	2.9-4.2	3.25	.862	.807×10 <sup>-6</sup>	.17 ×10 <sup>-3</sup>	.936×10 <sup>-6</sup>	.226×10 <sup>-3</sup>	.022
"	"	"	"	"	"	3.1	"	.665×10 <sup>-6</sup>	.171×10 <sup>-3</sup>	.771×10 <sup>-6</sup>	.227×10 <sup>-3</sup>	.046
"	"	"	"	"	"	3.01	"	.537×10 <sup>-6</sup>	.171×10 <sup>-3</sup>	.623×10 <sup>-6</sup>	"	.073
"	"	"	"	"	"	3.03	"	.57 ×10 <sup>-6</sup>	.171×10 <sup>-3</sup>	.661×10 <sup>-6</sup>	"	.066
"	"	"	"	.47×10 <sup>-3</sup>	2.9-4.2	2.96	1.044	.783×10 <sup>-6</sup>	.17 ×10 <sup>-3</sup>	.75 ×10 <sup>-6</sup>	.171×10 <sup>-3</sup>	.037
"	"	"	"	"	"	2.87	"	.62 ×10 <sup>-6</sup>	.171×10 <sup>-3</sup>	.594×10 <sup>-6</sup>	.172×10 <sup>-3</sup>	.067
"	"	"	"	"	"	2.87	"	.48 ×10 <sup>-6</sup>	.171×10 <sup>-3</sup>	.46 ×10 <sup>-6</sup>	"	.098
"	"	"	"	"	"	132.0	"	.14 ×10 <sup>-6</sup>	.172×10 <sup>-3</sup>	.134×10 <sup>-6</sup>	.173×10 <sup>-3</sup>	.295
"	"	"	"	"	"	2.85	"	.53 ×10 <sup>-6</sup>	.171×10 <sup>-3</sup>	.51 ×10 <sup>-6</sup>	.172×10 <sup>-3</sup>	.086
"	"	"	"	"	"	2.92	"	.711×10 <sup>-6</sup>	.170×10 <sup>-3</sup>	.68 ×10 <sup>-6</sup>	.171×10 <sup>-3</sup>	.049
"	"	----	"	.47×10 <sup>-3</sup>	2.9-4.0	2.82	1.03	.563×10 <sup>-6</sup>	.171×10 <sup>-3</sup>	.547×10 <sup>-6</sup>	.172×10 <sup>-3</sup>	.078
"	"	"	"	"	"	2.88	"	.718×10 <sup>-6</sup>	.170×10 <sup>-3</sup>	.697×10 <sup>-6</sup>	.171×10 <sup>-3</sup>	.047
"	"	"	"	"	"	2.87	"	.71 ×10 <sup>-6</sup>	.170×10 <sup>-3</sup>	.689×10 <sup>-6</sup>	.171×10 <sup>-3</sup>	.048
"	"	"	"	"	"	3.65	"	"	"	"	"	-.077
"	"	"	"	"	"	2.94	"	"	"	"	"	.033

80CB SAMPLE 2

f (Hz)	T <sub>ac</sub> (mK)	T <sub>dc</sub> (K)	T <sub>c</sub> (K)	D (J/K)	-log t	χ <sup>2</sup>	A <sup>+</sup> /A <sup>-</sup>	A <sup>+</sup> (J/K)	B <sup>+</sup> (J/K)	A <sup>-</sup> (J/K)	B <sup>-</sup> (J/K)	α
8.99	7	.385	340.209 ±.001 <sup>a</sup> , ±.015 <sup>b</sup>	.95×10 <sup>-3</sup>	3.0-4.0	8.88	1.16	-----	.353×10 <sup>-3</sup>	-----	.359×10 <sup>-3</sup>	.256
"	"	"	"	"	"	8.85	"	.636×10 <sup>-6</sup>	"	.548×10 <sup>-6</sup>	"	.257
"	"	"	"	"	"	8.76	"	.56 ×10 <sup>-6</sup>	.354×10 <sup>-3</sup>	.483×10 <sup>-6</sup>	.355×10 <sup>-3</sup>	.272
"	"	"	"	"	"	18.53	"	.337×10 <sup>-6</sup>	.338×10 <sup>-3</sup>	.291×10 <sup>-6</sup>	.341×10 <sup>-3</sup>	.097
"	"	"	"	"	2.8-3.8	8.09	"	.996×10 <sup>-6</sup>	.35 ×10 <sup>-3</sup>	.858×10 <sup>-6</sup>	.354×10 <sup>-3</sup>	.198
"	"	"	"	"	"	7.65	"	.726×10 <sup>-6</sup>	.353×10 <sup>-3</sup>	.626×10 <sup>-6</sup>	.357×10 <sup>-3</sup>	.24
"	"	"	"	"	"	10.0	"	.176×10 <sup>-5</sup>	.346×10 <sup>-3</sup>	.152×10 <sup>-5</sup>	.351×10 <sup>-3</sup>	.123
"	"	"	"	"	2.6-3.6	7.88	1.18	.115×10 <sup>-5</sup>	.35 ×10 <sup>-3</sup>	.975×10 <sup>-6</sup>	.354×10 <sup>-3</sup>	.178
"	"	"	"	"	"	7.5	"	.118×10 <sup>-5</sup>	.35 ×10 <sup>-3</sup>	.1 ×10 <sup>-5</sup>	"	.174
"	3-4	.179	"	.998×10 <sup>-3</sup>	2.8-4.6	4.42	1.08	-----	.35 ×10 <sup>-3</sup>	-----	.352×10 <sup>-3</sup>	.181
"	"	"	"	"	"	4.22	"	.104×10 <sup>-5</sup>	"	.963×10 <sup>-6</sup>	"	.185
"	"	"	"	"	"	4.13	"	"	"	"	"	.186
"	"	"	"	"	"	3.7	"	.1 ×10 <sup>-5</sup>	"	.923×10 <sup>-6</sup>	"	.191
"	"	"	"	"	"	3.2	"	.97 ×10 <sup>-6</sup>	"	.899×10 <sup>-6</sup>	"	.194
"	1-2	.1	340.193 ±.001 <sup>a</sup> , ±.015 <sup>b</sup>	.4 ×10 <sup>-3</sup>	2.4-5.0	2.67	1.2	.107×10 <sup>-5</sup>	.349×10 <sup>-3</sup>	.892×10 <sup>-6</sup>	.352×10 <sup>-3</sup>	.189
"	"	"	"	"	"	2.03	"	.103×10 <sup>-5</sup>	.35 ×10 <sup>-3</sup>	.858×10 <sup>-6</sup>	.353×10 <sup>-3</sup>	.194
"	"	"	"	"	"	2.02	"	.101×10 <sup>-5</sup>	"	.842×10 <sup>-6</sup>	"	.197
"	"	"	"	"	2.4-4.6	2.24	"	.9 ×10 <sup>-6</sup>	"	.75 ×10 <sup>-6</sup>	.354×10 <sup>-3</sup>	.211
"	"	"	"	"	"	1.96	"	.846×10 <sup>-6</sup>	"	.71 ×10 <sup>-6</sup>	"	.219
"	"	"	"	"	"	1.69	"	.617×10 <sup>-6</sup>	.353×10 <sup>-3</sup>	.514×10 <sup>-6</sup>	.357×10 <sup>-3</sup>	.257
"	"	"	"	"	"	1.56	"	.559×10 <sup>-6</sup>	.353×10 <sup>-3</sup>	.466×10 <sup>-6</sup>	.357×10 <sup>-3</sup>	.271
"	"	"	"	"	2.4-4.4	1.9	"	.589×10 <sup>-6</sup>	.353×10 <sup>-3</sup>	.49 ×10 <sup>-6</sup>	.36 ×10 <sup>-6</sup>	.264
"	"	"	"	"	"	1.66	"	.559×10 <sup>-6</sup>	"	.466×10 <sup>-6</sup>	"	.27
"	"	"	"	"	"	1.58	"	.81 ×10 <sup>-6</sup>	.351×10 <sup>-3</sup>	.675×10 <sup>-6</sup>	.358×10 <sup>-3</sup>	.225
"	"	"	"	"	2.9-4.8	2.63	"	.66 ×10 <sup>-6</sup>	.353×10 <sup>-3</sup>	.55 ×10 <sup>-6</sup>	.359×10 <sup>-3</sup>	.245
"	"	"	"	"	"	2.14	"	.85 ×10 <sup>-6</sup>	.35 ×10 <sup>-3</sup>	.71 ×10 <sup>-6</sup>	.35 ×10 <sup>-3</sup>	.216
"	"	"	"	"	"	"	"	.988×10 <sup>-6</sup>	"	.82 ×10 <sup>-6</sup>	"	.199



f (Hz)	T <sub>ac</sub> (mK)	T <sub>dc</sub> (K)	T <sub>c</sub> (K)	D (J/K)	-log t	χ <sup>2</sup>	A <sup>+</sup> /A <sup>-</sup>	A <sup>+</sup> (J/K)	B <sup>+</sup> (J/K)	A <sup>-</sup> (J/K)	B <sup>-</sup> (J/K)	α
3.17	2-3	≤.1	336.486 ±.001a, ±.05b	.3×10 <sup>-3</sup>	2.5-4.0	2.12	.575	.495×10 <sup>-5</sup>	.199×10 <sup>-3</sup>	.861×10 <sup>-5</sup>	.185×10 <sup>-3</sup>	-.156
"	"	"	"	"	"	2.14	"	.475×10 <sup>-5</sup>	"	.826×10 <sup>-5</sup>	"	-.149
"	"	"	"	"	"	2.08	"	----	.197×10 <sup>-3</sup>	----	.181×10 <sup>-3</sup>	-.178
"	"	"	"	"	"	2.67	"	----	.205×10 <sup>-3</sup>	----	.195×10 <sup>-3</sup>	-.06
"	"	"	"	"	2.2-3.6	2.26	.566	.483×10 <sup>-5</sup>	.199×10 <sup>-3</sup>	.853×10 <sup>-5</sup>	.185×10 <sup>-3</sup>	-.152

f (Hz)	T <sub>ac</sub> (mK)	T <sub>dc</sub> (K)	T <sub>c</sub> (K)	D (J/K)	-log t	χ <sup>2</sup>	A <sup>+</sup> /A <sup>-</sup>	A <sup>+</sup> (J/K)	B <sup>+</sup> (J/K)	A <sup>-</sup> (J/K)	B <sup>-</sup> (J/K)	α
8.97	3-4	206	356.096 ±.001 <sup>a</sup>	.35×10 <sup>-3</sup>	2.6-4.2	9.0	.744	.125×10 <sup>-5</sup>	.303×10 <sup>-3</sup>	.168×10 <sup>-6</sup>	.301×10 <sup>-3</sup>	-.22
"	"	"	"	"	"	9.2	"	.272×10 <sup>-6</sup>	.305×10 <sup>-3</sup>	.366×10 <sup>-6</sup>	.304×10 <sup>-3</sup>	.0065
"	"	"	"	"	"	9.2	"	.178×10 <sup>-6</sup>	.306×10 <sup>-3</sup>	.239×10 <sup>-6</sup>	.305×10 <sup>-3</sup>	.068
"	"	"	"	"	2.7-3.8	9.14	.742	.44 ×10 <sup>-6</sup>	.305×10 <sup>-3</sup>	.593×10 <sup>-6</sup>	.304×10 <sup>-3</sup>	-.069
"	"	"	"	"	"	9.17	"	.412×10 <sup>-6</sup>	.305×10 <sup>-3</sup>	.555×10 <sup>-6</sup>	.304×10 <sup>-3</sup>	-.059
"	"	"	"	"	"	9.19	"	.27 ×10 <sup>-6</sup>	.306×10 <sup>-3</sup>	.364×10 <sup>-6</sup>	.305×10 <sup>-3</sup>	.001
"	"	"	"	"	"	9.2	"	.24 ×10 <sup>-6</sup>	.306×10 <sup>-3</sup>	.323×10 <sup>-6</sup>	"	.016
"	"	"	"	"	"	9.19	"	.257×10 <sup>-6</sup>	.306×10 <sup>-3</sup>	.346×10 <sup>-6</sup>	"	.0075
"	"	"	"	"	"	9.23	"	.376×10 <sup>-6</sup>	.305×10 <sup>-3</sup>	.507×10 <sup>-6</sup>	.304×10 <sup>-3</sup>	-.047
8.97	3-4	161	356.099 ±.001 <sup>a</sup>	.35×10 <sup>-3</sup>	2.6-4.0	6.45	.870	.3 ×10 <sup>-6</sup>	.307×10 <sup>-3</sup>	.35 ×10 <sup>-6</sup>	.306×10 <sup>-3</sup>	.05
"	"	"	"	"	"	7.55	"	.1 ×10 <sup>-6</sup>	.308×10 <sup>-3</sup>	.11 ×10 <sup>-6</sup>	.401×10 <sup>-3</sup>	.195
"	"	"	"	"	"	6.26	"	.108×10 <sup>-5</sup>	.305×10 <sup>-3</sup>	.12 ×10 <sup>-5</sup>	.304×10 <sup>-3</sup>	-.124
"	"	"	"	"	"	6.65	"	.7 ×10 <sup>-6</sup>	.306×10 <sup>-3</sup>	.80 ×10 <sup>-6</sup>	.305×10 <sup>-3</sup>	-.066
"	"	"	"	"	"	7.1	"	.456×10 <sup>-6</sup>	.307×10 <sup>-3</sup>	.524×10 <sup>-6</sup>	.306×10 <sup>-3</sup>	-.005
"	"	"	"	"	2.6-3.7	6.79	.763	.366×10 <sup>-6</sup>	.307×10 <sup>-3</sup>	.48 ×10 <sup>-6</sup>	.306×10 <sup>-3</sup>	.023
"	"	"	"	"	"	7.5	"	.158×10 <sup>-6</sup>	.308×10 <sup>-3</sup>	.207×10 <sup>-6</sup>	.307×10 <sup>-3</sup>	.140
"	"	"	"	"	"	8.0	"	.215×10 <sup>-6</sup>	.308×10 <sup>-3</sup>	.282×10 <sup>-6</sup>	"	.097
"	"	"	"	"	"	7.6	"	.911×10 <sup>-6</sup>	.306×10 <sup>-3</sup>	.119×10 <sup>-5</sup>	.305×10 <sup>-3</sup>	-.108

CBOOA (continued)

f (Hz)	T <sub>ac</sub> (mK)	T <sub>dc</sub> (K)	T <sub>c</sub> (K)	D (J/K)	-log t	χ <sup>2</sup>	A <sup>+</sup> /A <sup>-</sup>	A <sup>+</sup> (J/K)	B <sup>+</sup> (J/K)	A <sup>-</sup> (J/K)	B <sup>-</sup> (J/K)	α
8.97	3-4	713	356.950 ±.001a	.35×10 <sup>-3</sup>	2.2-4.0	7.8	.786	.292×10 <sup>-5</sup>	.259×10 <sup>-3</sup>	.369×10 <sup>-5</sup>	.257×10 <sup>-3</sup>	-.26
"	"	"	"	"	2.5-3.8	7.8	.719	.906×10 <sup>-6</sup>	.264×10 <sup>-3</sup>	.126×10 <sup>-5</sup>	.263×10 <sup>-3</sup>	-.077
"	"	"	"	"	"	4.7	"	.166×10 <sup>-5</sup>	.262×10 <sup>-3</sup>	.231×10 <sup>-5</sup>	.260×10 <sup>-3</sup>	-.159
"	"	"	"	"	2.2-3.6	4.76	.778	.124×10 <sup>-5</sup>	.263×10 <sup>-3</sup>	.159×10 <sup>-5</sup>	.262×10 <sup>-3</sup>	-.124
"	"	"	"	"	"	5.6	"	.428×10 <sup>-6</sup>	.265×10 <sup>-3</sup>	.550×10 <sup>-6</sup>	.265×10 <sup>-3</sup>	.045
"	"	"	"	"	"	7.2	"	.458×10 <sup>-6</sup>	.265×10 <sup>-3</sup>	.589×10 <sup>-6</sup>	"	.033
"	"	"	"	"	"	9.5	"	.375×10 <sup>-6</sup>	.265×10 <sup>-3</sup>	.482×10 <sup>-6</sup>	"	.064
"	"	"	"	"	"	12.7	"	.472×10 <sup>-6</sup>	.265×10 <sup>-3</sup>	.607×10 <sup>-6</sup>	"	.029
8.97	5-6	1,210	356.95 ±.001a	.33×10 <sup>-3</sup>	2.3-3.7	5.88	1.213	.203×10 <sup>-5</sup>	.288×10 <sup>-3</sup>	.167×10 <sup>-5</sup>	.29 ×10 <sup>-3</sup>	-.109
"	"	"	"	"	"	6.24	"	.144×10 <sup>-5</sup>	.29 ×10 <sup>-3</sup>	.119×10 <sup>-5</sup>	.292×10 <sup>-3</sup>	-.057
"	"	"	"	"	"	14.54	"	.133×10 <sup>-6</sup>	.295×10 <sup>-3</sup>	.110×10 <sup>-6</sup>	.296×10 <sup>-3</sup>	.299
"	"	"	"	"	2.4-3.5	5.05	1.13	.974×10 <sup>-6</sup>	.292×10 <sup>-3</sup>	.862×10 <sup>-6</sup>	.293×10 <sup>-3</sup>	-.0067
"	"	"	"	"	"	5.09	"	.98 ×10 <sup>-6</sup>	.29 ×10 <sup>-3</sup>	.867×10 <sup>-6</sup>	.291×10 <sup>-3</sup>	-.008
"	"	"	"	"	"	4.99	"	.15 ×10 <sup>-5</sup>	.29 ×10 <sup>-3</sup>	.133×10 <sup>-5</sup>	"	-.07
"	"	"	"	"	2.4-3.4	5.18	1.12	.948×10 <sup>-6</sup>	.292×10 <sup>-3</sup>	.846×10 <sup>-6</sup>	.292×10 <sup>-3</sup>	-.002

f (Hz)	T <sub>ac</sub> (mK)	T <sub>dc</sub> (K)	T <sub>c</sub> (K)	D (J/K)	-log t	χ <sup>2</sup>	A <sup>+</sup> /A <sup>-</sup>	A <sup>+</sup> (J/K)	B <sup>+</sup> (J/K)	A <sup>-</sup> (J/K)	B <sup>-</sup> (J/K)	α
1.43	1	.053	307.017 ±.001a, ±.012b	.5×10 <sup>-2</sup>	2.6-4.2	7.29	.963	.127×10 <sup>-4</sup>	.161×10 <sup>-2</sup>	.132×10 <sup>-4</sup>	.163×10 <sup>-2</sup>	.325
"	"	"	"	"	"	7.55	"	.162×10 <sup>-4</sup>	.158×10 <sup>-2</sup>	.168×10 <sup>-4</sup>	.160×10 <sup>-2</sup>	.295
"	"	"	"	"	"	7.79	"	.204×10 <sup>-4</sup>	.154×10 <sup>-2</sup>	.212×10 <sup>-4</sup>	.156×10 <sup>-2</sup>	.267
"	"	"	"	"	"	8.0	"	----	.154×10 <sup>-2</sup>	----	.159×10 <sup>-2</sup>	.289
"	"	"	"	"	2.6-4.0	4.96	.971	.166×10 <sup>-4</sup>	.158×10 <sup>-2</sup>	.171×10 <sup>-4</sup>	.16 ×10 <sup>-2</sup>	.291
"	"	"	"	"	"	4.92	"	.179×10 <sup>-4</sup>	.157×10 <sup>-2</sup>	.179×10 <sup>-4</sup>	.159×10 <sup>-2</sup>	.284
"	"	"	"	"	"	4.99	"	----	----	----	----	.268
"	3	.101	307.016 ±001a, ±.012b	"	2.6-4.2	12.18	1.13	.147×10 <sup>-5</sup>	.37 ×10 <sup>-3</sup>	.131×10 <sup>-5</sup>	.39 ×10 <sup>-3</sup>	.424
"	"	"	"	"	"	17.75	"	.101×10 <sup>-5</sup>	.378×10 <sup>-3</sup>	.894×10 <sup>-6</sup>	.397×10 <sup>-3</sup>	.471
"	"	"	"	"	"	15.8	1.13	.139×10 <sup>-5</sup>	.371×10 <sup>-3</sup>	.123×10 <sup>-5</sup>	.391×10 <sup>-3</sup>	.431
"	"	"	"	"	2.6-4.5	20.0	1.13	.177×10 <sup>-5</sup>	.365×10 <sup>-3</sup>	.157×10 <sup>-5</sup>	.383×10 <sup>-3</sup>	.4
"	"	"	"	"	"	19.5	"	.123×10 <sup>-5</sup>	.374×10 <sup>-3</sup>	.109×10 <sup>-5</sup>	.391×10 <sup>-3</sup>	.45
"	"	"	"	"	"	20.74	"	.126×10 <sup>-5</sup>	.373×10 <sup>-3</sup>	.112×10 <sup>-5</sup>	.39 ×10 <sup>-3</sup>	.44
"	"	"	"	"	3.0-4.0	8.5	"	.143×10 <sup>-5</sup>	.37 ×10 <sup>-3</sup>	.127×10 <sup>-5</sup>	.388×10 <sup>-3</sup>	.421
"	"	"	"	"	"	11.9	"	.242×10 <sup>-5</sup>	.356×10 <sup>-3</sup>	.214×10 <sup>-5</sup>	.375×10 <sup>-3</sup>	.361
"	"	"	"	"	"	13.0	"	.147×10 <sup>-5</sup>	.37 ×10 <sup>-3</sup>	.13 ×10 <sup>-5</sup>	.388×10 <sup>-3</sup>	.422
"	"	"	"	"	"	12.3	"	.181×10 <sup>-5</sup>	.365×10 <sup>-3</sup>	.16 ×10 <sup>-5</sup>	.384×10 <sup>-3</sup>	.397

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25. Chromel A is a registered trademark of Hoskins Manufacturing Co., Detroit, Michigan, which is used for a nickel-chromium wire that they manufacture.
26. Torr Seal is a registered trademark of Varian in Palo Alto, California, and is a low vapor pressure epoxy made by mixing equal lengths of resin and hardener. It is cured at room temperature in 30 minutes.
27. Epo-Tek 417 is a two component, electrically conductive epoxy containing pure silver; 1/15 parts by weight of hardener is added to the silver base and the epoxy is cured for two hours at 80° C. Epo-Tek 417 is made by Epoxy Technology, Inc., in Billerica, MA.
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